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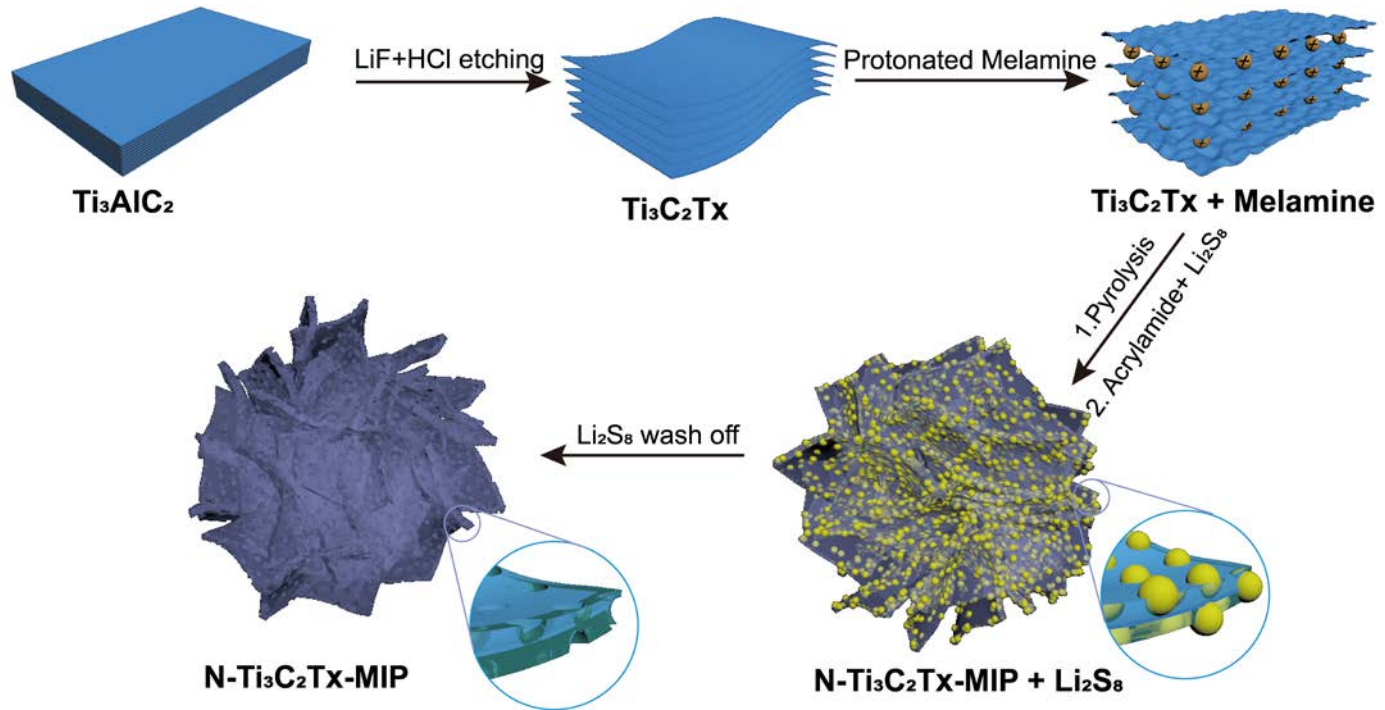
X-ray photoelectron spectroscopy: how does it work and what can we learn from it?

Petra Rudolf

Zernike Institute for Advanced Materials, Groningen

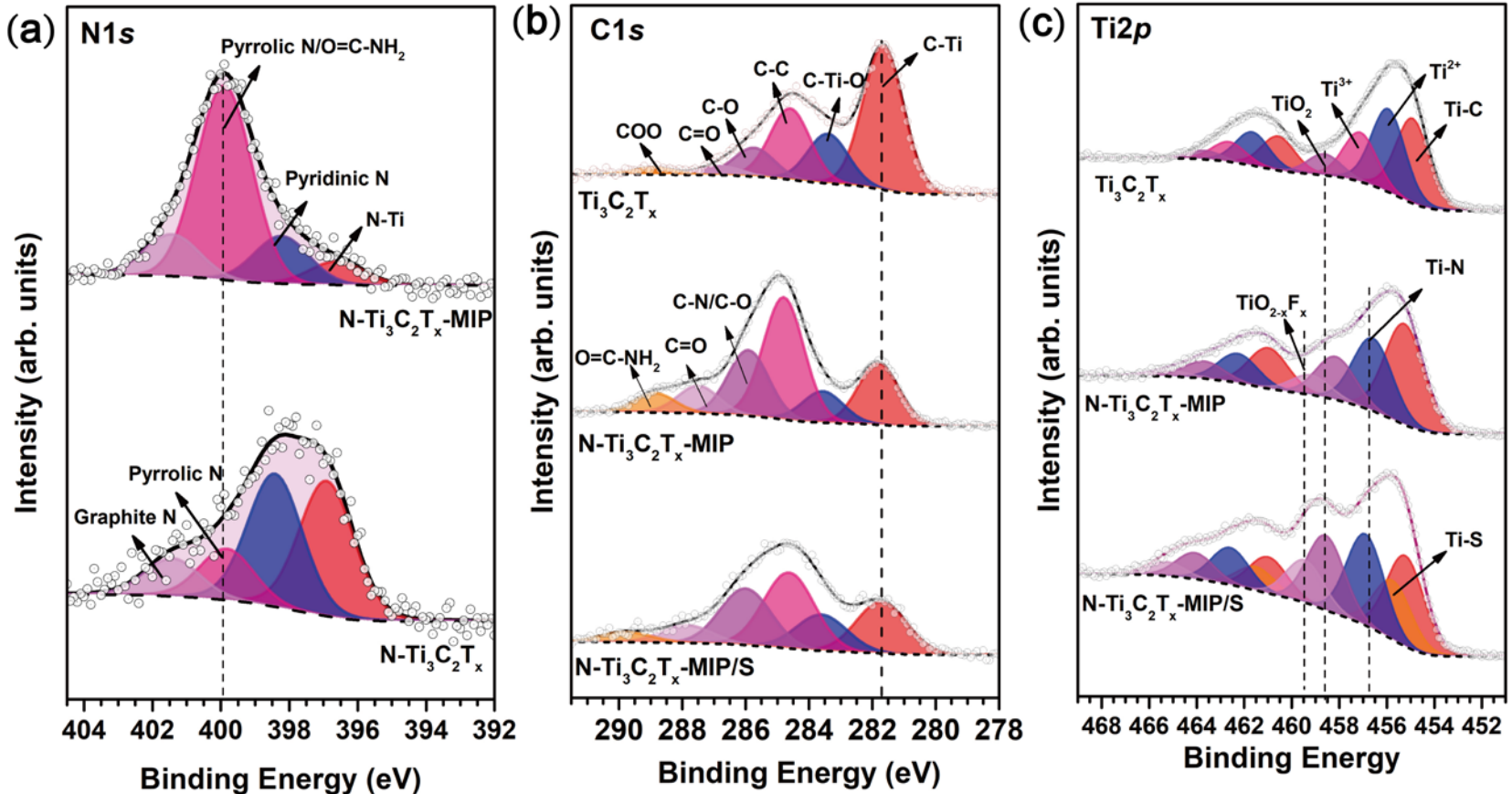


Molecularly imprinted polymer (MIP) in conjunction with 2D material, MXene, developed for cathodes of Li-S batteries



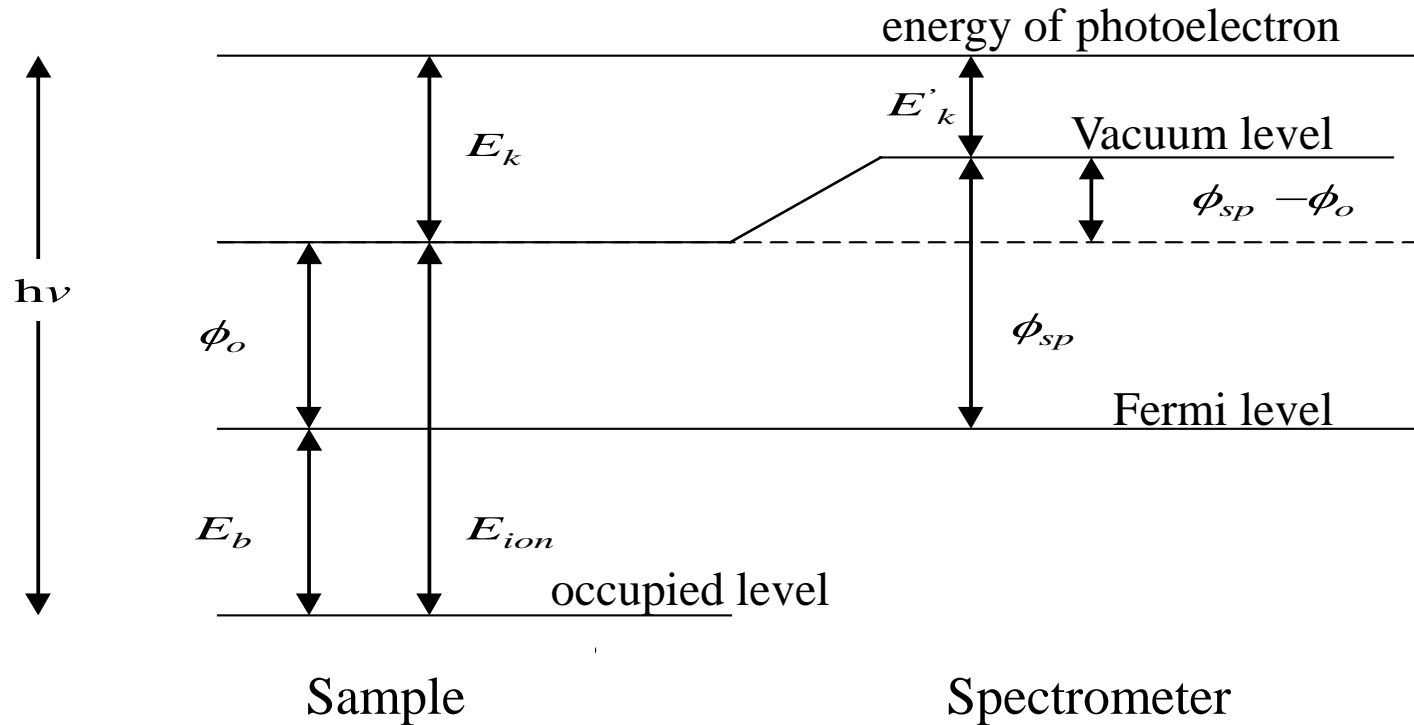


Molecularly imprinted polymer (MIP) in conjunction with 2D material, MXene, developed for cathodes of Li-S batteries





What is the binding energy in XPS ?



$$h\nu = E_b + E_k + \phi_o$$

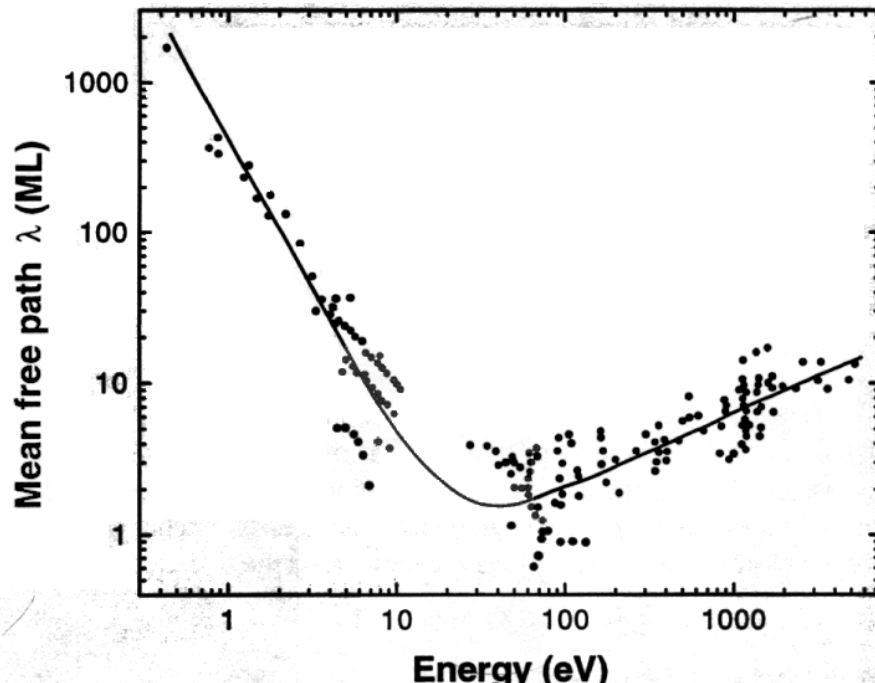
$$E_b = E_f - E_i = h\nu - E'_k + \phi_{sp}$$



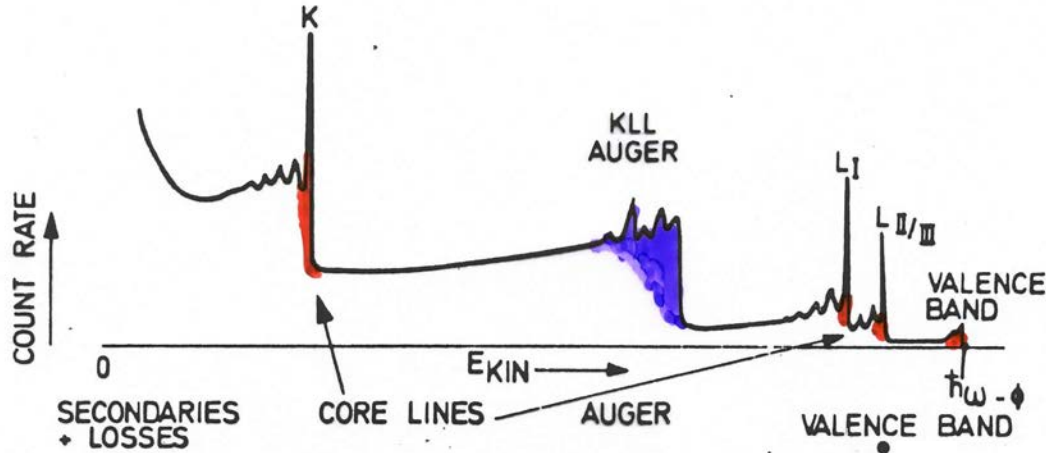
Photoemission can be seen as 3 step process

- 1) **photoionisation** - optical absorption machinery, selection rules
- 2) **transport** of photoelectron through sample - inelastic mean free path
- 3) **emission** of photoelectron into vacuum - refractive effects at the surface, k -parallel vs. k -normal

Universal curve of electron mean free path (Zangwill ' 88)



$E_k < 1500$ eV : XPS
gives information on
surface layer (typically
<10 nm)



The photoionisation of core level
→ Auger decay or fluorescent
decay (X-ray emission)

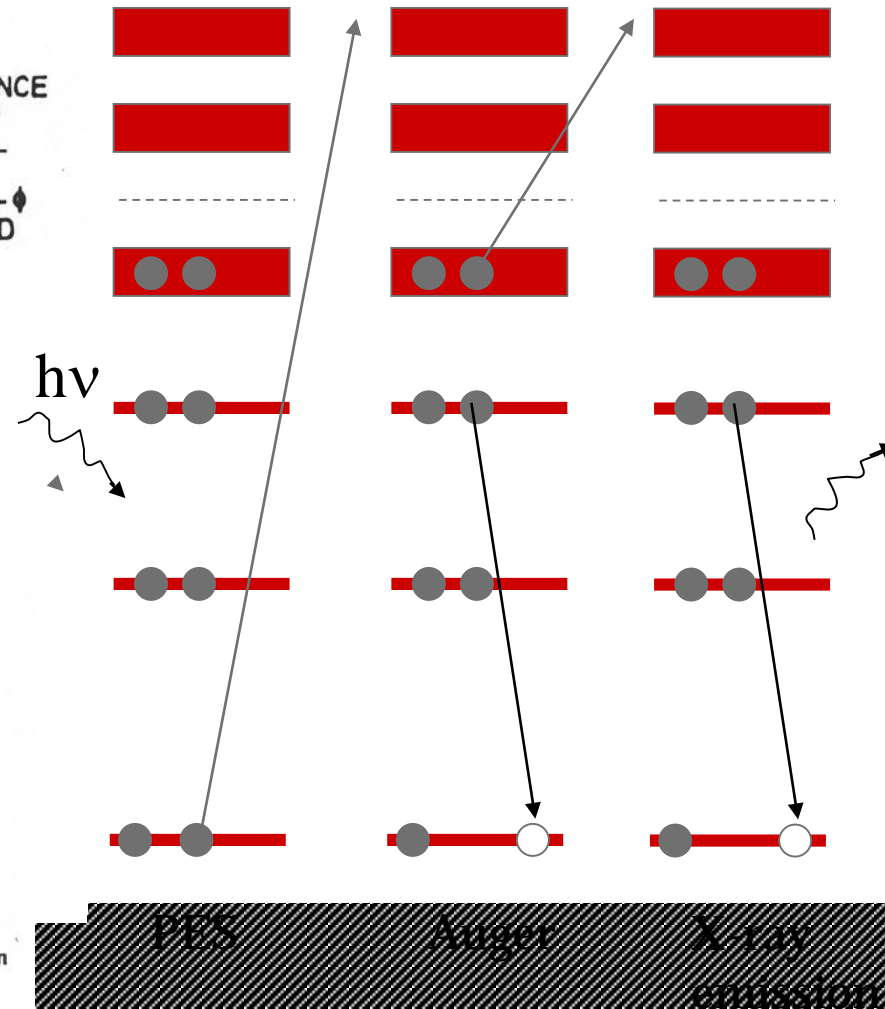
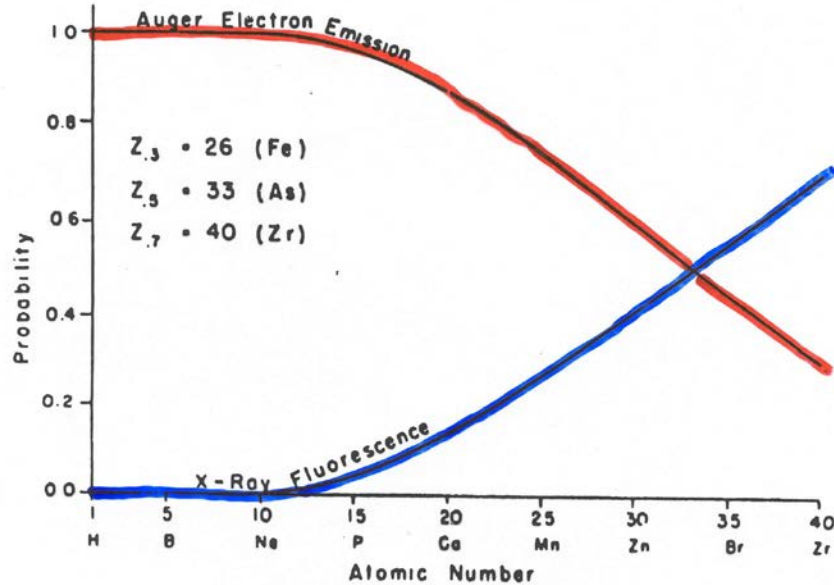
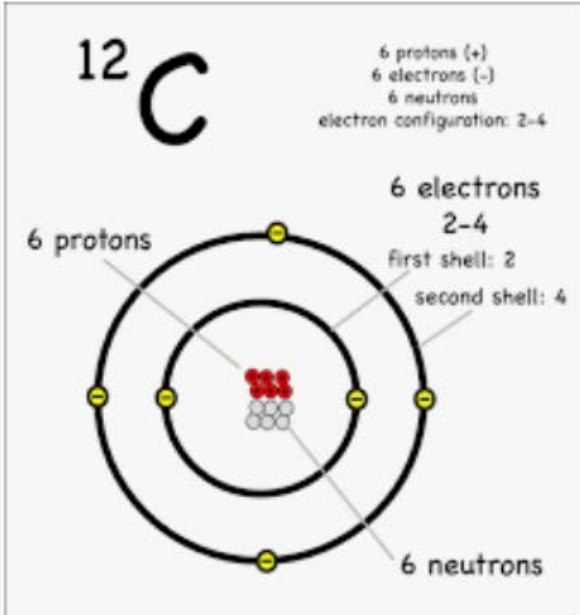


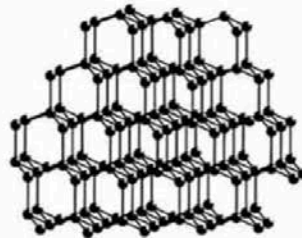
Figure 3.5 Relative probabilities of relaxation by emission of an Auger electron and by emission of an X-ray photon of characteristic energy, following creation of a core hole in the K shell



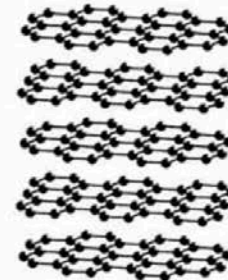
Let's start simple: Carbon



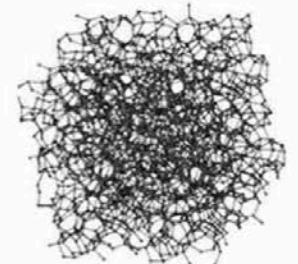
DIAMOND



GRAPHITE

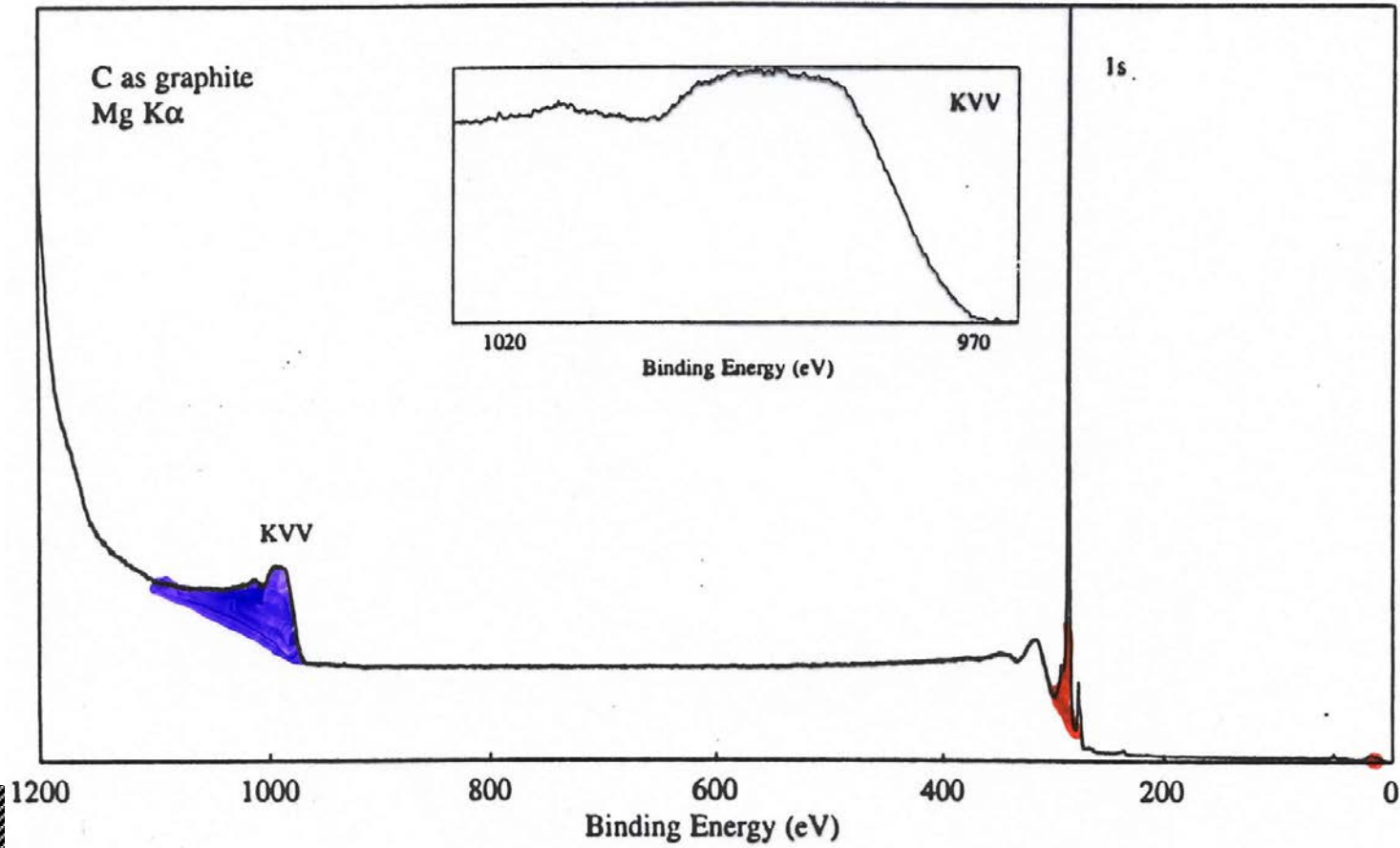


AMORPHOUS CARBON



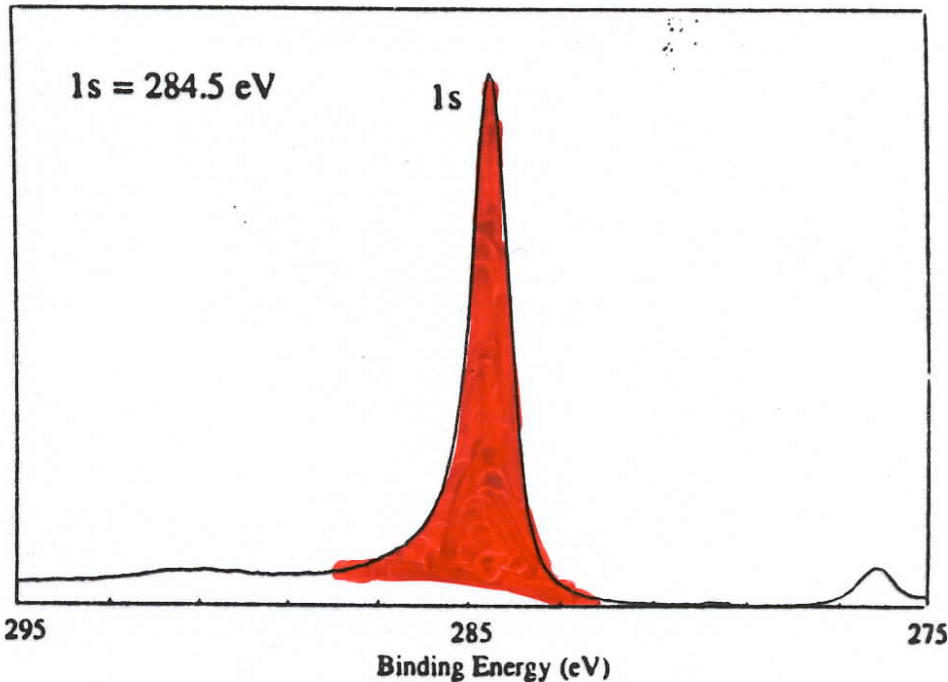


Wide XPS spectrum of graphite (C)





Core level XPS spectrum of graphite (C)



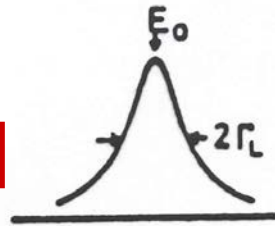
The singlet **C 1s** line is characterized by:

- 1) A specific binding energy which reflects the specific atomic species (C) in a specific chemical environment
- 2) A finite width reflecting the instrumental resolution, lifetime broadening and other many-body effects

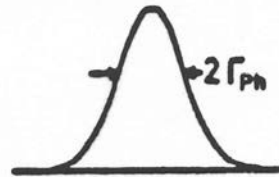


What determines the line shape ?

Lorentzian Lifetime broadening



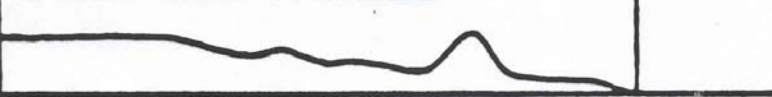
Gaussian Phonon broadening



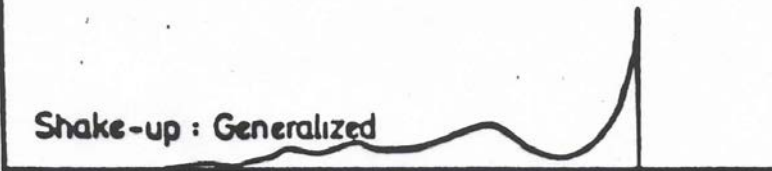
Extrinsic losses : Plasmons



Extrinsic losses : Generalized



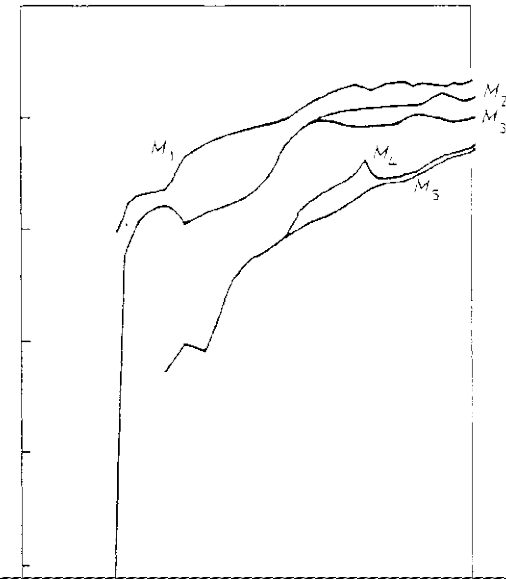
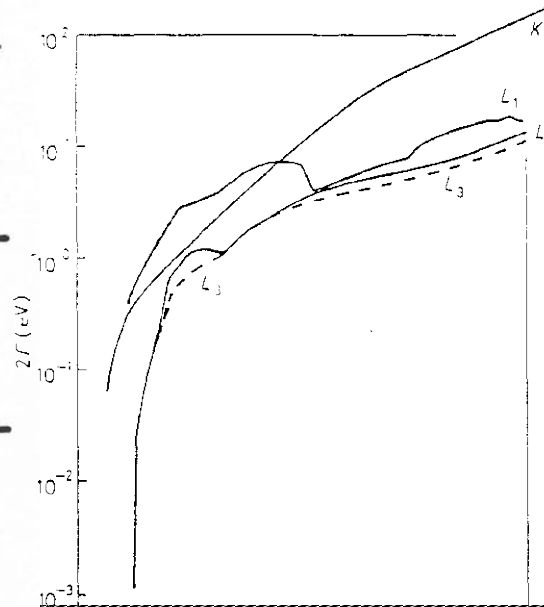
Shake-up : Generalized



← K. E. →

photoionisation process takes 10^{-17} sec – core hole of light element decays in 10^{-15} sec – indetermination in energy

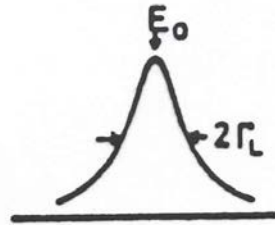
(4) $\frac{1}{2}$ FWHM = $\Gamma_L = \hbar/\tau = 6.58 \cdot 10^{-16}/\tau$ eV,



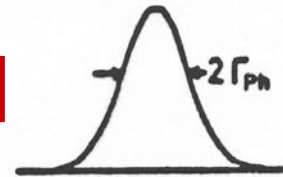


What determines the line shape ?

Lorentzian Lifetime broadening



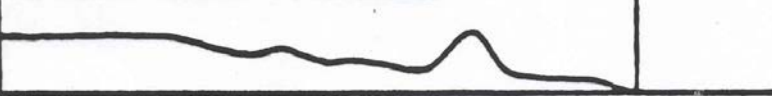
Gaussian Phonon broadening



Extrinsic losses : Plasmons



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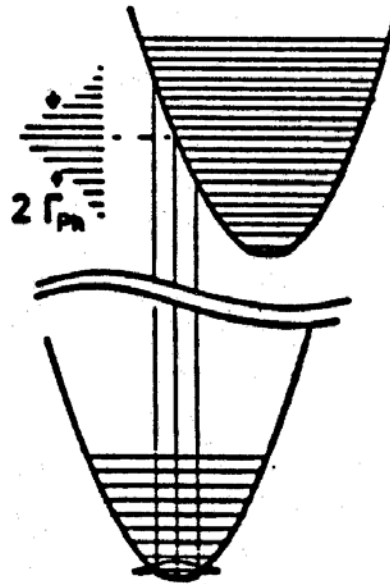
Shake-up : Generalized



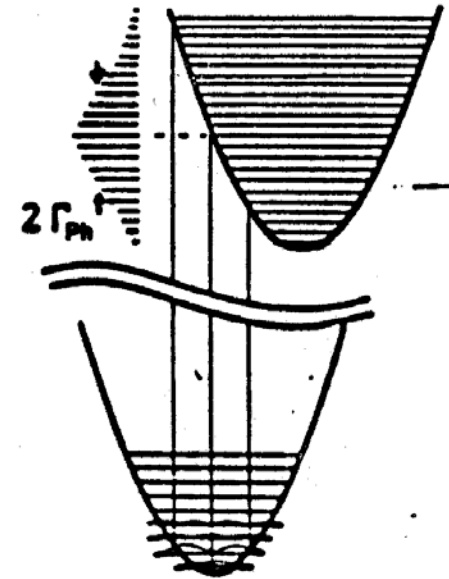
— K. E. —>

VIBRATIONAL BROADENING

(a) $T=0\text{ K}$



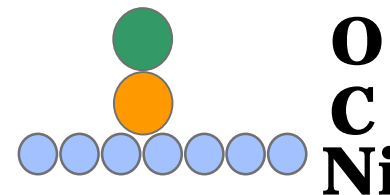
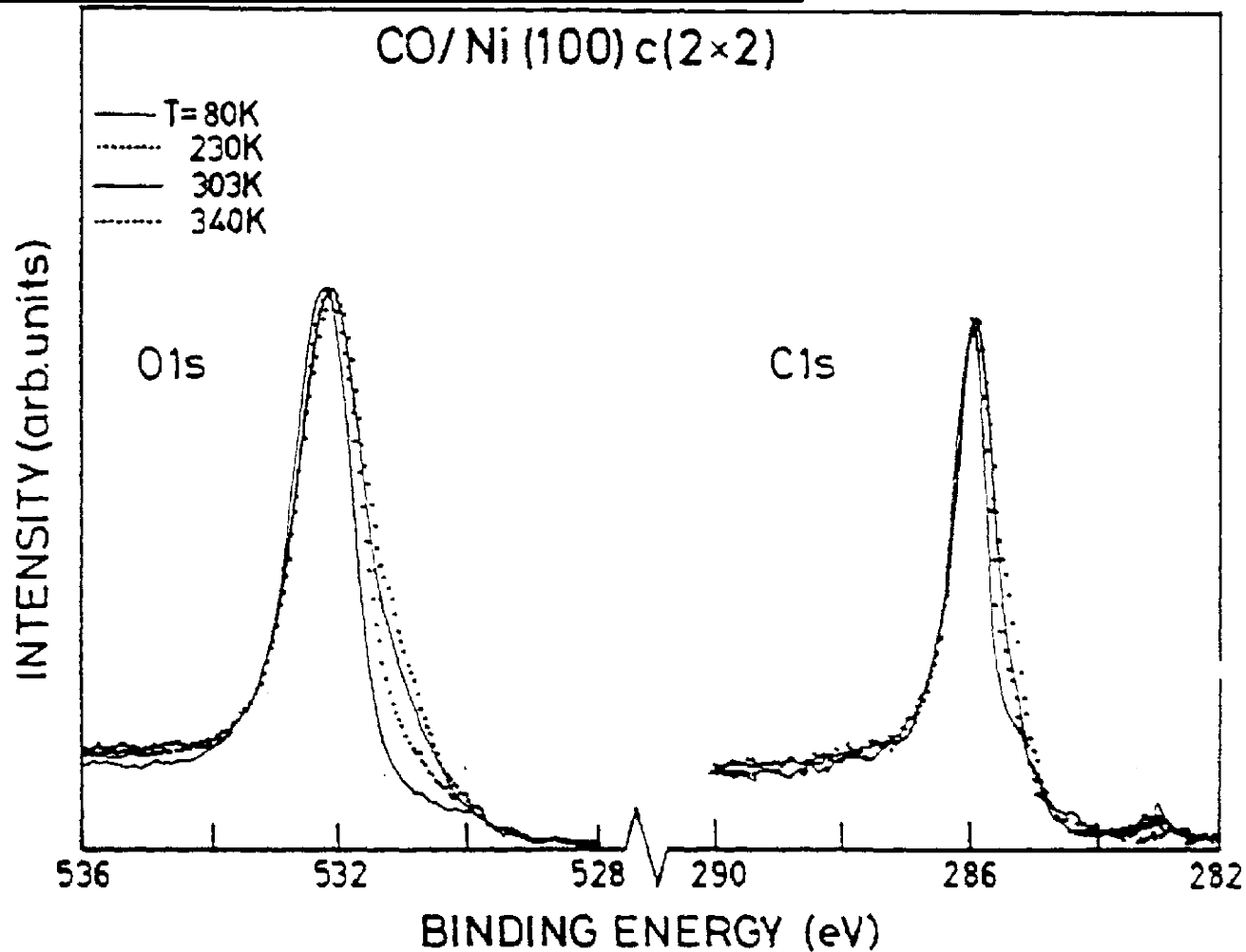
(b) $T>0\text{ K}$



The origin of phonon broadening in XPS and Auger lines.



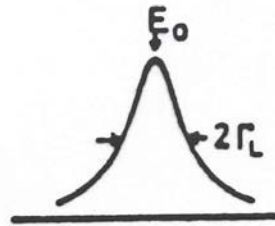
Vibrational Broadening



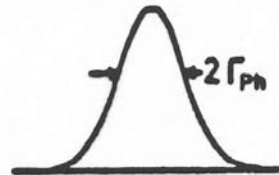


What determines the line shape ?

Lorentzian Lifetime broadening



Gaussian Phonon broadening



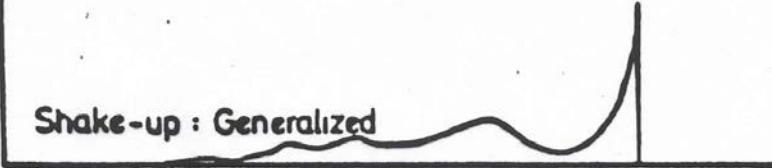
Extrinsic losses : Plasmons



Extrinsic losses : Generalized

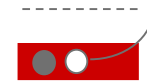
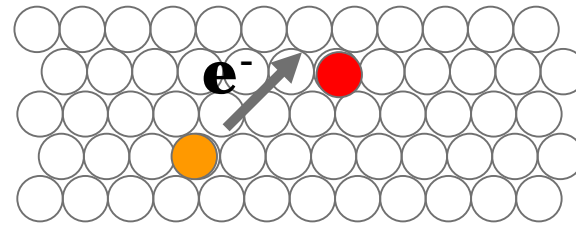


Shake-up : Generalized



K. E. →

Extrinsic losses



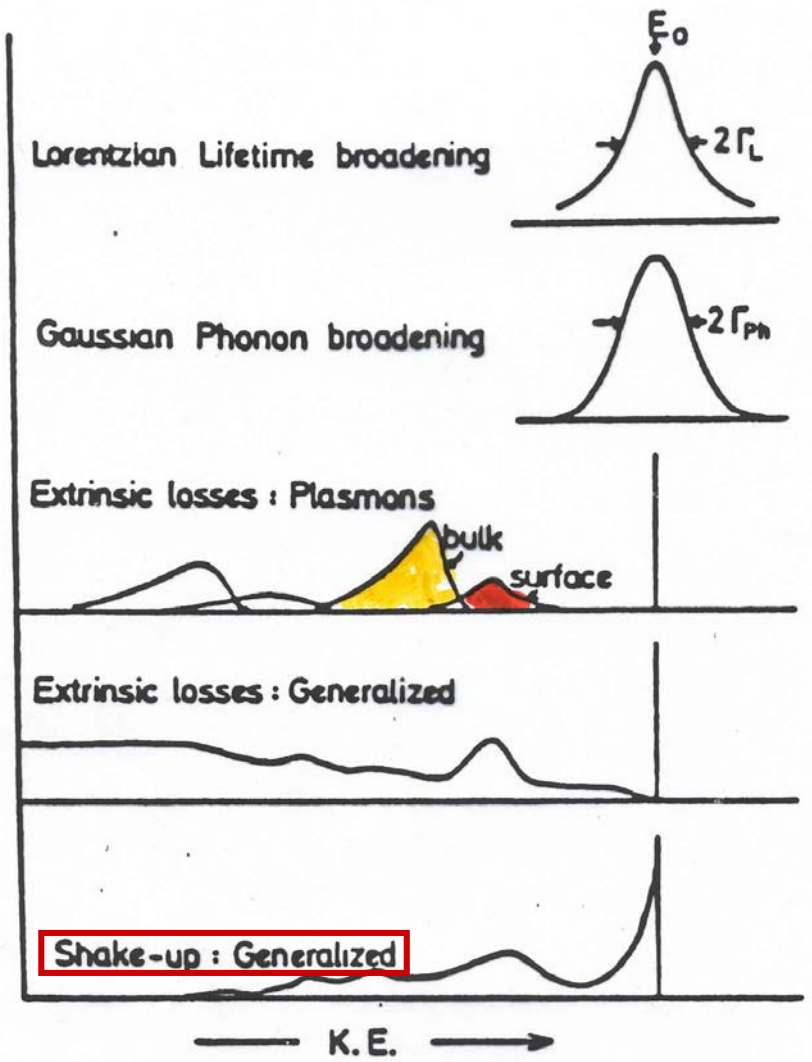
• single electron transitions

• collective excitations = plasmons

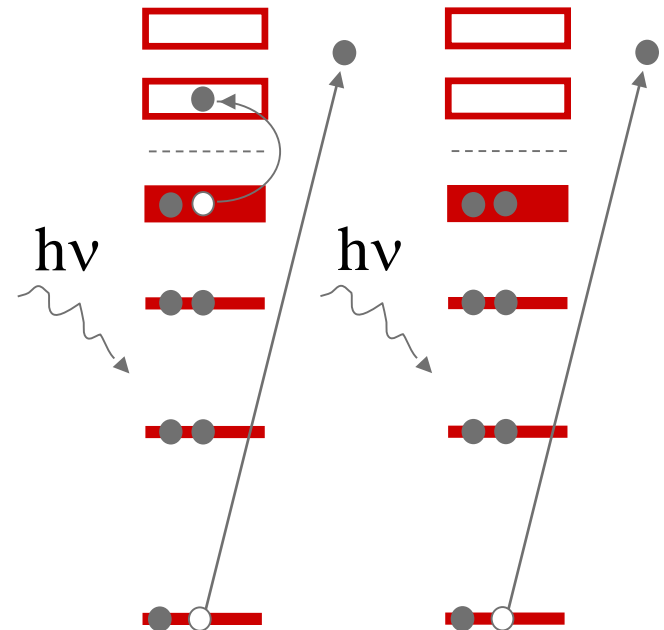
photoelectron loses energy
spectral features appear at lower kinetic energy



What determines the line shape ?



Shake-up: photoemission is not a single particle process - different final states possible





Line-asymmetry in Core Level XPS

The line-asymmetry scales with the density of states at E_F

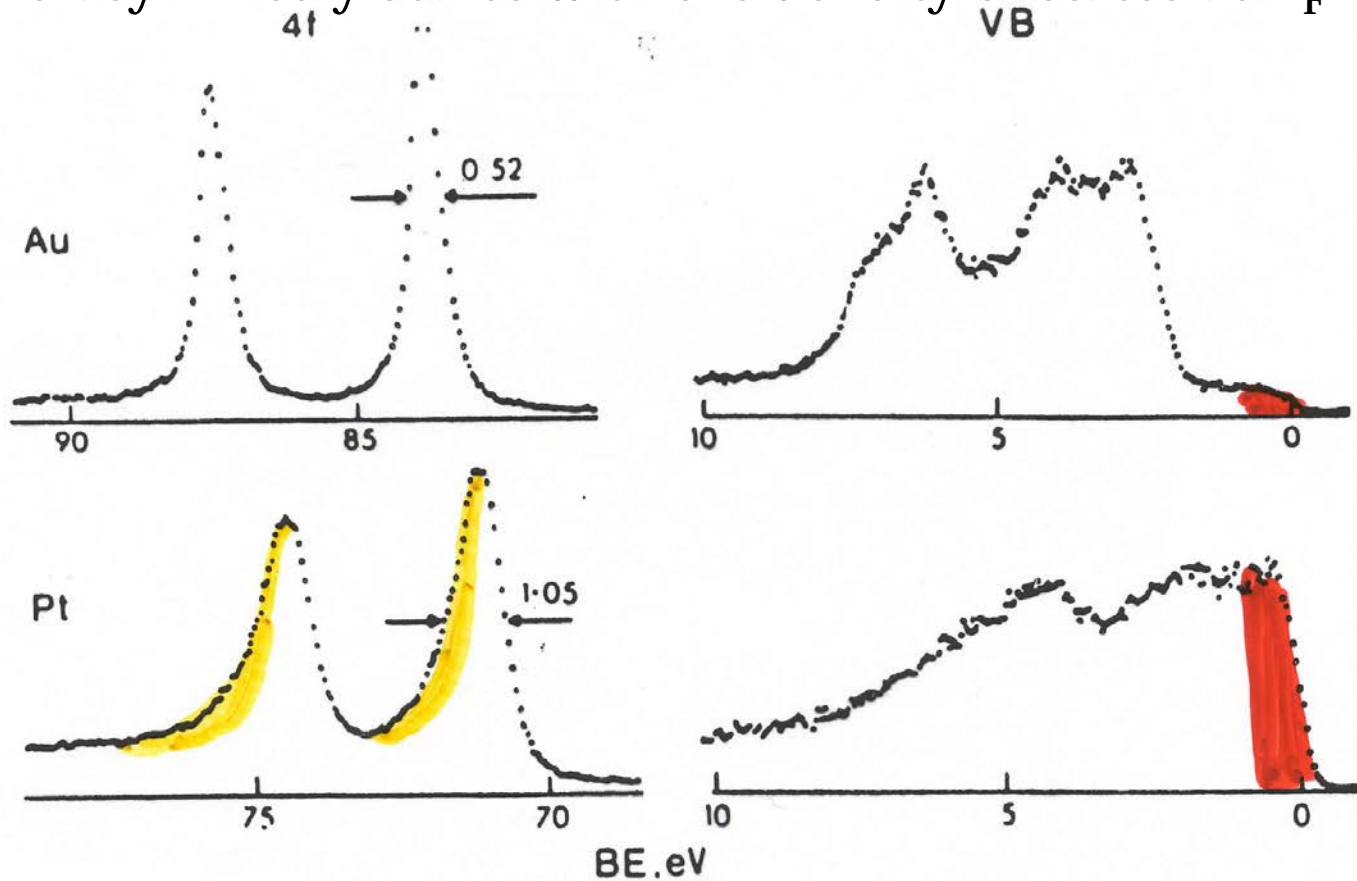
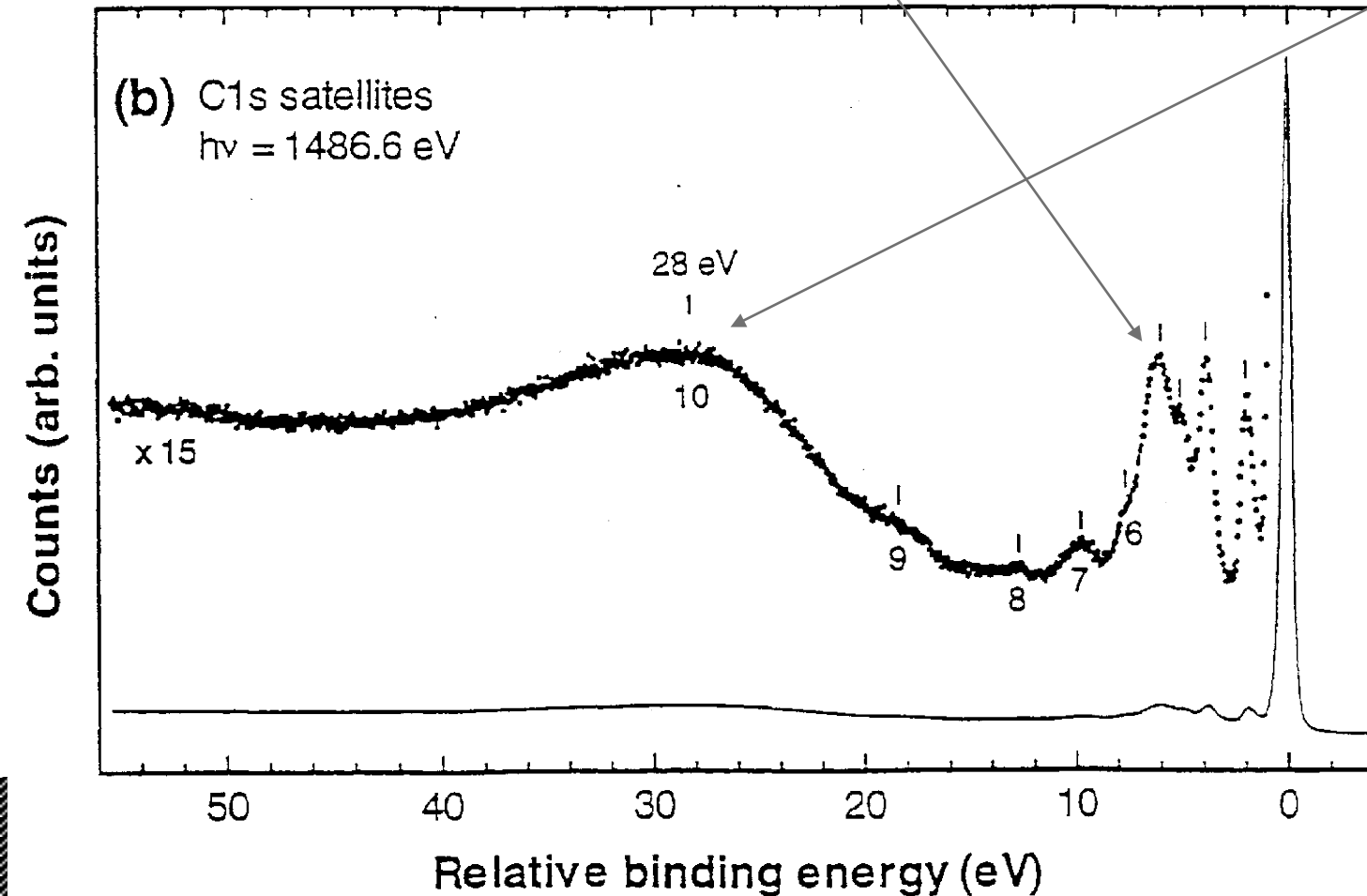


Figure 3.33 Core (4f) and valence (VB) photoelectron spectra of gold and platinum recorded using monochromatic Al K α radiation. Note the relationship between the degree of core level asymmetry and the density of states at the Fermi level (BE = 0 eV). (After Barrie, Swift and Briggs⁴⁹)



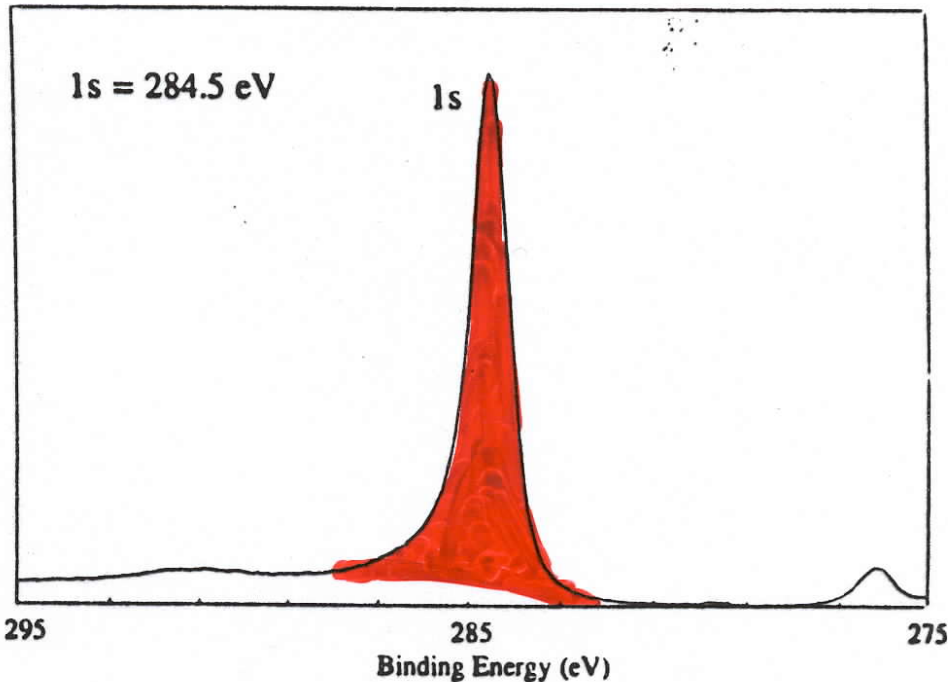
Example: C_{60} : 3 valence electrons of carbon form σ orbitals – cage structure,
 1 valence electron in π orbitals.

single electron excitations; all $\pi \rightarrow \pi^*$: π plasmon; all $\pi + \sigma \rightarrow \pi^* + \sigma^*$: $\sigma + \pi$ plasmon





Core level XPS spectrum of graphite (C)

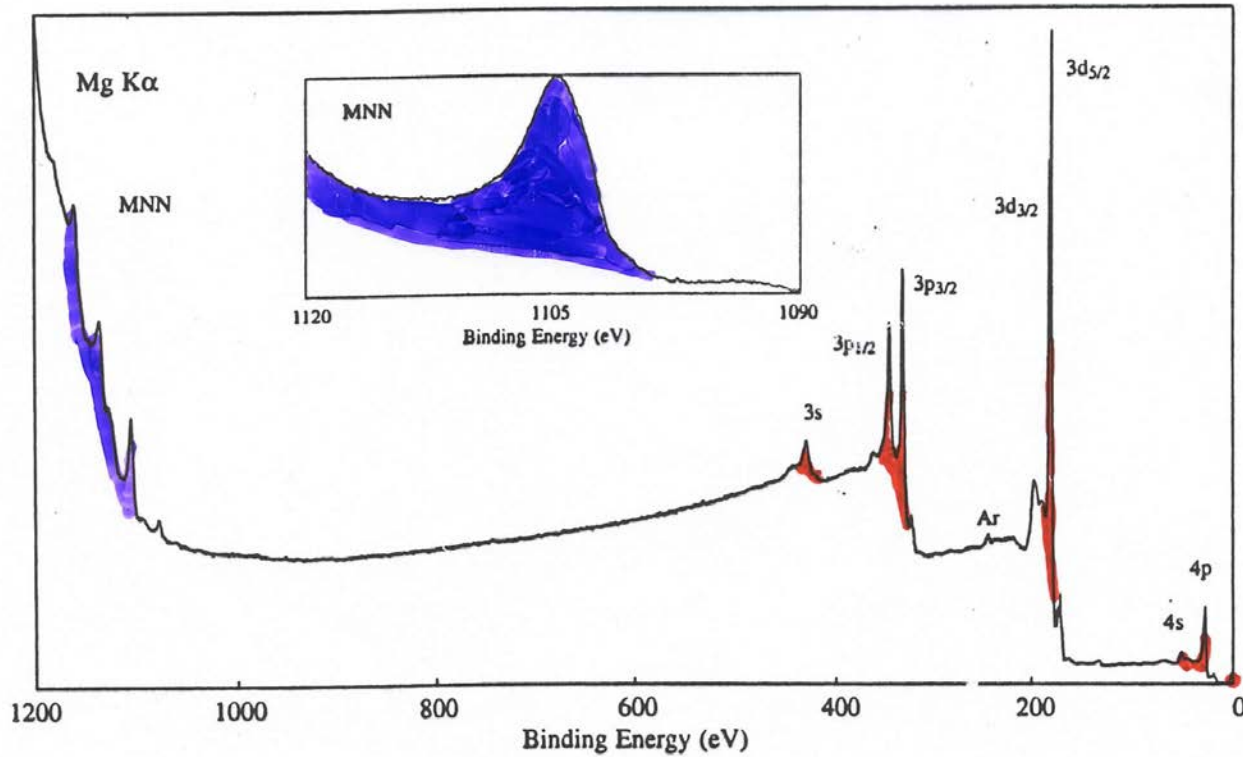


The singlet **C 1s** line is characterized by:

- 1) A specific binding energy which reflects the specific atomic species (C) in a specific chemical environment
- 2) A finite width reflecting the instrumental resolution, lifetime broadening and other many-body effects

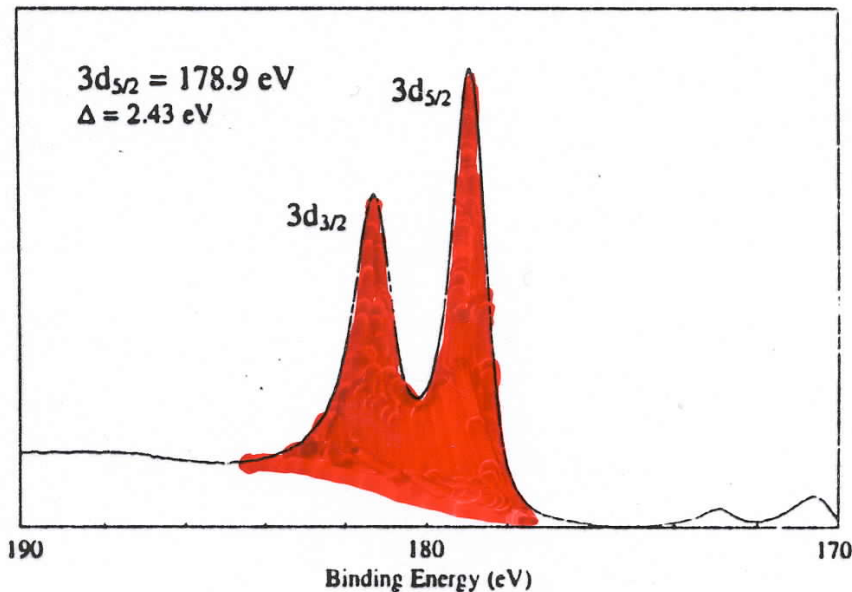


Wide XPS spectrum of zirconium (Zr)





Core level XPS spectrum of zirconium (Zr)



The Zr 3d line is characterized by:

- 1) A specific binding energy which reflects the specific atomic species (Zr) in a specific chemical environment
- 2) **The occurrence of a doublet reflecting the core level spin-orbit splitting**
- 3) A finite width reflecting the instrumental resolution, lifetime broadening and other many-body effects



Spin-Orbit Splitting

Quantum Numbers

- j Total Angular Momentum
- l Orbital Angular Momentum
- s Spin Angular Momentum

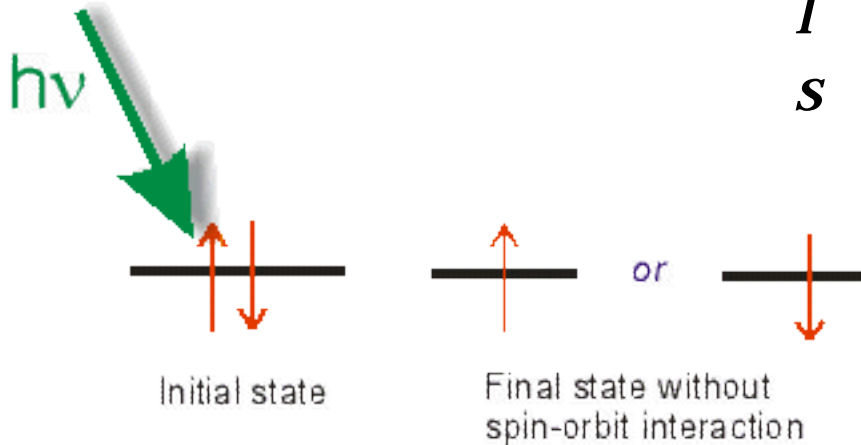
$$j = l + s$$

p-symmetry state

$$l = 1$$

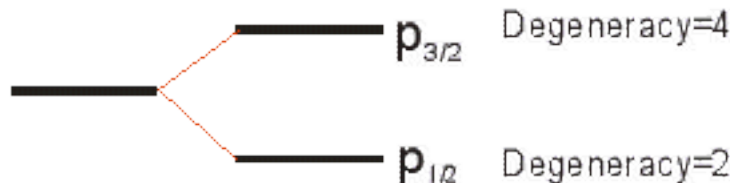
$$s = \pm 1/2$$

(i)



(ii)

Spin orbit interaction lifts degeneracy

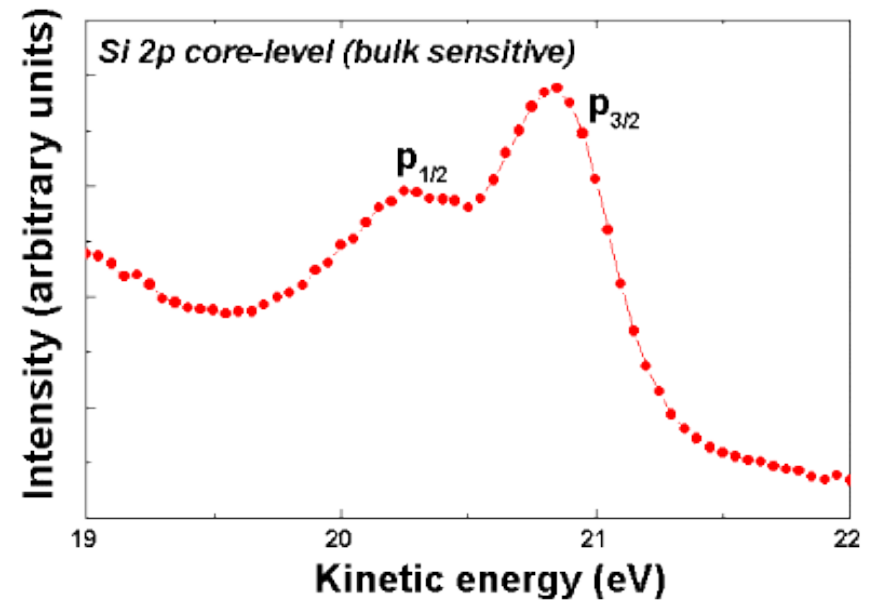
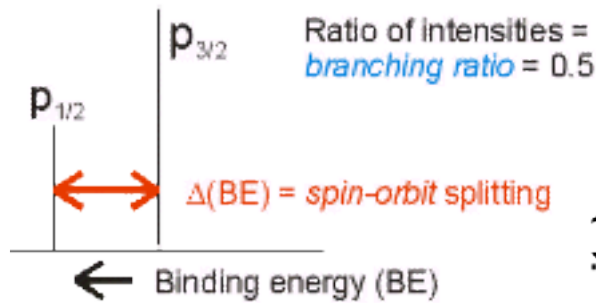


$$\text{Degeneracy} = |2j+1|$$



Spin-Orbit Splitting

(iii) Degeneracies determine relative intensities of peaks comprising doublet



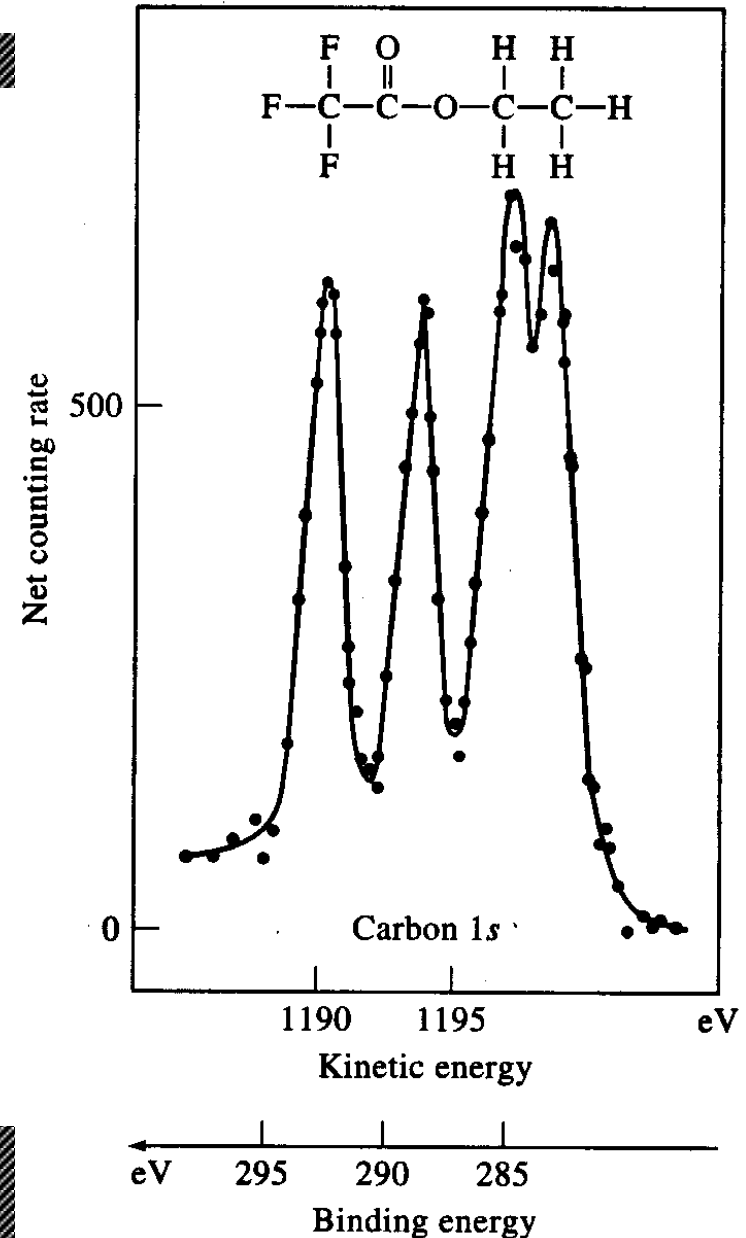


XPS : Chemical shift

Carbon 1s binding energies in ethyl trifluoroacetate

Can be used to quantify different chemical bonds for same element

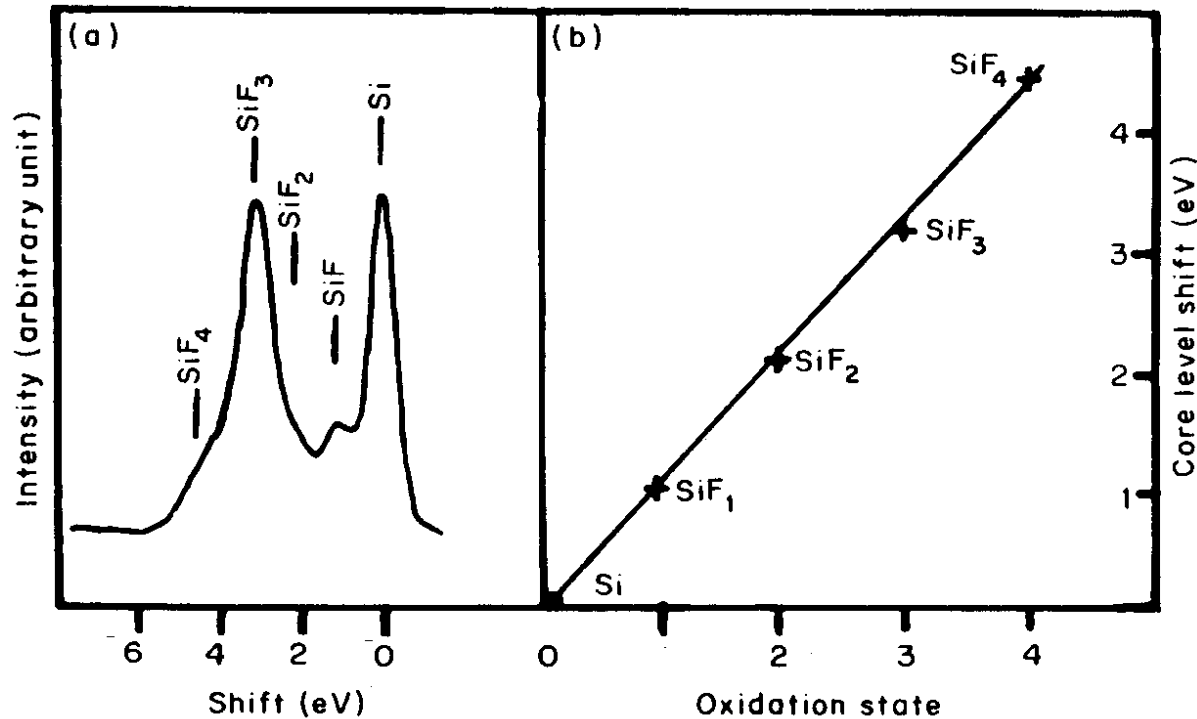
XPS also called **ESCA** -
 electron spectroscopy
 for chemical analysis





XPS : Chemical shift

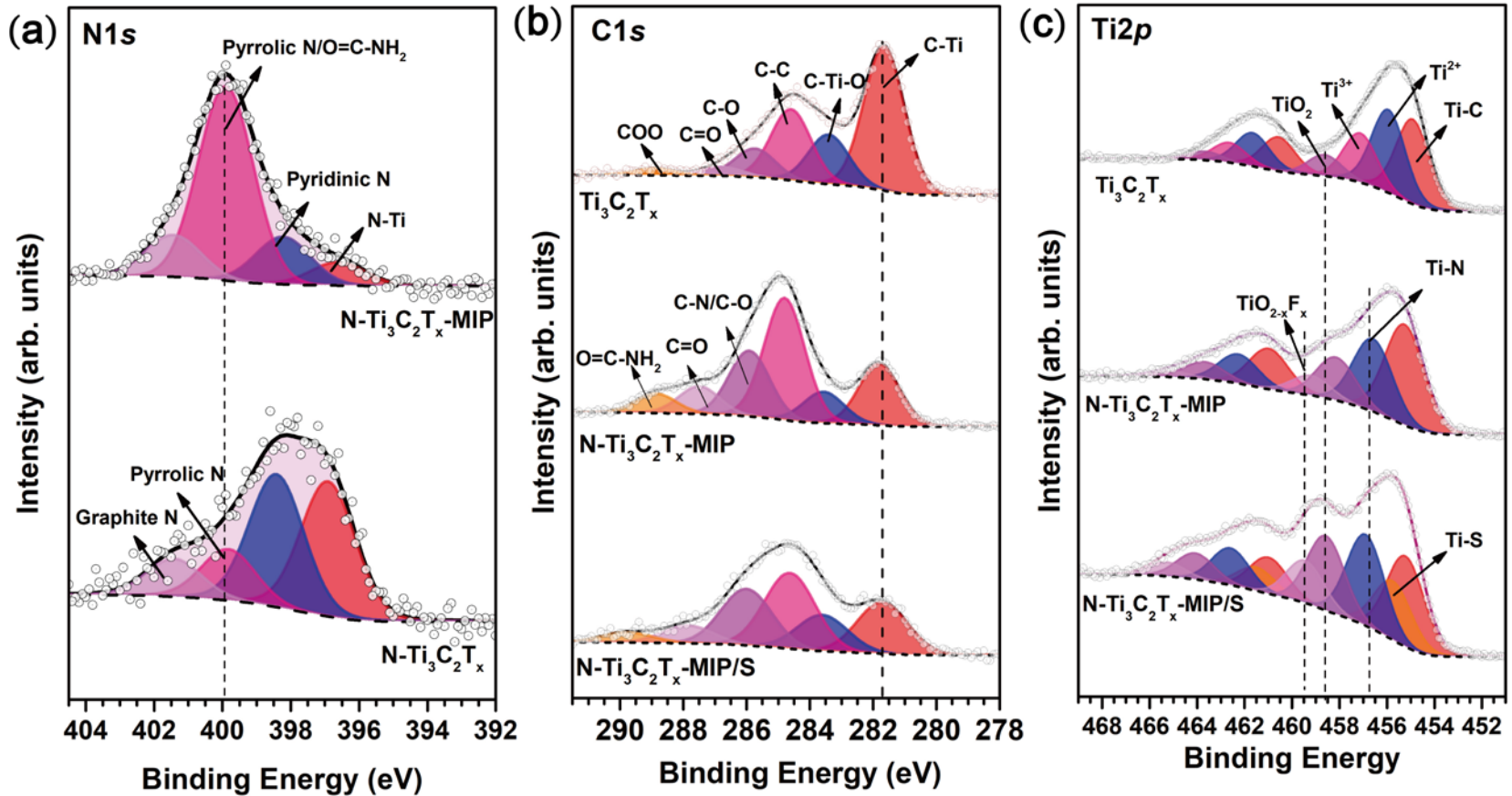
Si 2p spectra for a fluorine-etched silicon wafer

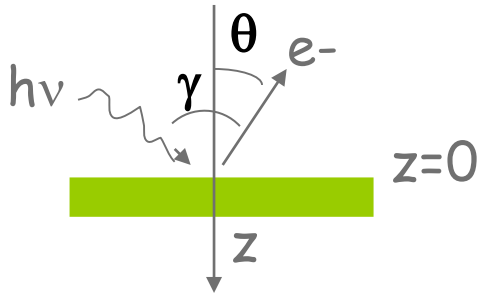


binding energies increase as the oxidation state increases - can be used to determine oxidation state



Molecularly imprinted polymer (MIP) in conjunction with 2D material, MXene, developed for cathodes of Li-S batteries





Once the photon flux Φ is given, the photoelectron intensity I_i of the (nl) orbital of the i -th atomic species is approximately given by

$$I_i(nl) = C_i \lambda(E_{kin}) \phi(\hbar\omega) \sigma_{nl}(\hbar\omega) T(E_{kin})$$

Where Φ is the photon flux

C_i Atomic Concentration of the i -th species

λ Escape Depth

σ_{nl} Orbital Cross Section

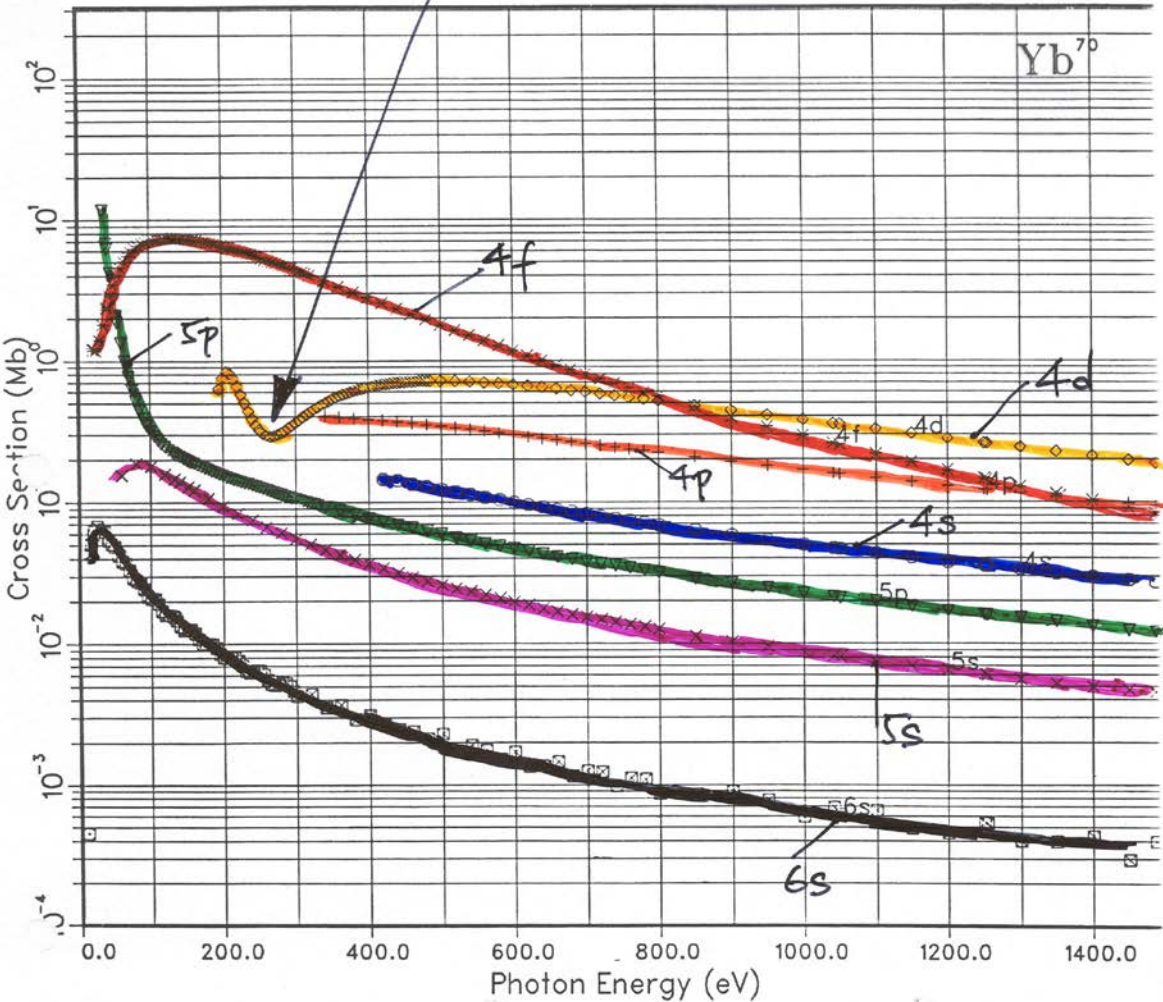
T Instrumental Efficiency

sensitivity factor $S_i = \sigma_i(h\nu) T(E_{kin})$



COOPER MINIMUM

Yb⁷⁰



Calculated photoionization cross sections for free atoms vs. photon energy (Yeh and Lindau)

Yb binding energies(eV) are:

| | | |
|----------------|----------------|----------------|
| 1s(2) 57009.8 | 2s(2) 9290.78 | 2p(6) 8897.20 |
| 3s(2) 2086.76 | 3p(6) 1904.14 | 4s(2) 402.610 |
| 3d(10) 1561.01 | 4p(6) 327.461 | 5s(2) 50.8637 |
| 4d(10) 190.652 | 5p(6) 30.7582 | 6s(2) 5.16038 |
| 4f(14) 15.9818 | | |



Appendix E. Atomic Sensitivity Factors for X-ray Sources at 90°

This table is based upon empirical peak area values corrected for the system's transmission function. The values are only valid for and should only be applied when the electron energy analyzer used has the transmission characteristics of the spherical capacitor type analyzer equipped with an Omni Focus III lens supplied by Perkin-Elmer. The data are calculated for x-rays at 90° relative to the analyzer.*

| Element | Line | ASF | Element | Line | ASF | Element | Line | ASF | Element | Line | ASF |
|---------|-------------------|-------|---------|-------------------|-------|---------|-------------------|-------|---------|-------------------|-------|
| Ag | 3d | 5.198 | Eu | 4d | 2.210 | Na | 1s | 1.685 | Si | 2p | 0.283 |
| Al | 2p | 0.193 | F | 1s | 1.000 | Nb | 3d | 2.517 | Sm | 3d _{5/2} | 2.907 |
| Ar | 2p | 1.011 | Fe | 2p | 2.686 | Nd | 3d | 4.697 | Sn | 3d _{5/2} | 4.095 |
| As | 3d | 0.570 | Ga | 2p _{3/2} | 3.341 | Ne | 1s | 1.340 | Sr | 3d | 1.578 |
| Au | 4f | 5.240 | Gd | 4d | 2.207 | Ni | 2p | 3.653 | Ta | 4f | 2.589 |
| B | 1s | 0.159 | Ge | 2p _{3/2} | 3.100 | O | 1s | 0.711 | Tb | 4d | 2.201 |
| Ba | 4d | 2.627 | Hf | 4f | 2.221 | Os | 4f | 3.747 | Tc | 3d | 3.266 |
| Bc | 1s | 0.074 | Hg | 4f | 5.797 | P | 2p | 0.412 | Te | 3d _{5/2} | 4.925 |
| Bi | 4f | 7.632 | Ho | 4d | 2.189 | Pb | 4f | 6.968 | Tb | 4f _{7/2} | 7.498 |
| Br | 3d | 0.895 | I | 3d _{5/2} | 5.337 | Pd | 3d | 4.642 | Ti | 2p | 1.798 |
| C | 1s | 0.296 | In | 3d _{5/2} | 3.777 | Pm | 3d | 3.754 | Tl | 4f | 6.447 |
| Ca | 2p | 1.634 | Ir | 4f | 4.217 | Pr | 3d | 6.356 | Tm | 4d | 2.172 |
| Cd | 3d _{5/2} | 3.444 | K | 2p | 1.300 | Pt | 4f | 4.674 | U | 4f _{7/2} | 8.476 |
| Ce | 3d | 7.399 | Kr | 3d | 1.096 | Rb | 3d | 1.316 | V | 2p | 1.912 |
| Cl | 2p | 0.770 | La | 3d | 7.708 | Re | 4f | 3.327 | W | 4f | 2.959 |
| Co | 2p | 3.255 | Li | 1s | 0.025 | Rh | 3d | 4.179 | Xe | 3d _{5/2} | 5.702 |
| Cr | 2p | 2.201 | Lu | 4d | 2.156 | Ru | 3d | 3.696 | Y | 3d | 1.867 |
| Cs | 3d _{5/2} | 6.032 | Mg | 2s | 0.252 | S | 2p | 0.570 | Yb | 4d | 2.169 |
| Cu | 2p | 4.798 | Mn | 2p | 2.420 | Sb | 3d _{5/2} | 4.473 | Zn | 2p _{3/2} | 3.354 |
| Dy | 4d | 2.198 | Mo | 3d | 2.867 | Sc | 2p | 1.678 | Zr | 3d | 2.216 |
| Er | 4d | 2.184 | N | 1s | 0.477 | Se | 3d | 0.722 | | | |

Provided by
 the maker of
 the
 spectrometer



$$I_i(nl) = C_i \lambda(E_{kin}) \phi(\hbar\omega) \sigma_{nl}(\hbar\omega) T(E_{kin})$$

Once the **efficiency of detection of an atomic species** is calibrated via the **sensitivity factors** one gets

Homogeneous binary compound :

$$\frac{I_A}{I_B} = \frac{S_A N_A \lambda(E_A)}{S_B N_B \lambda(E_B)}$$

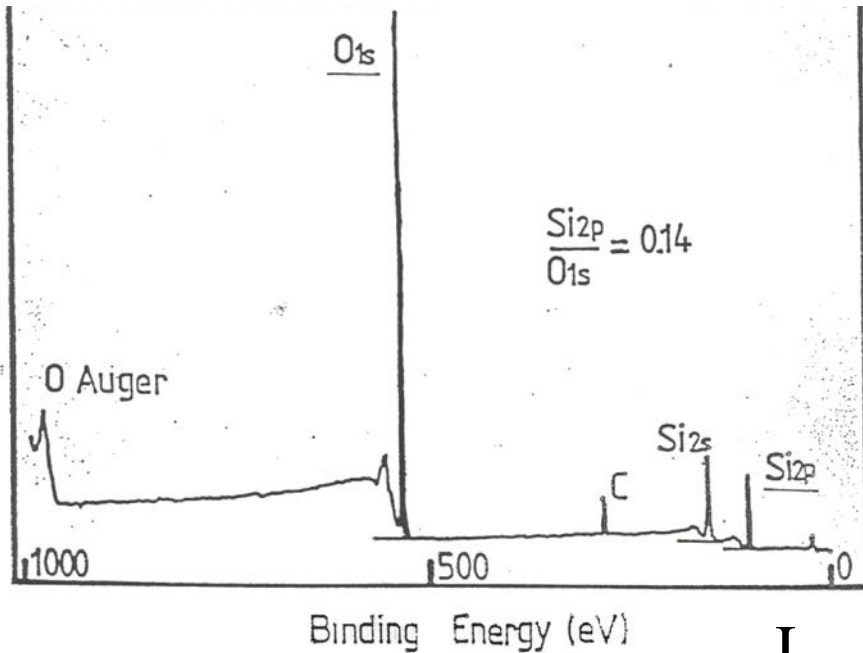
Sample with many components :

Atomic concentration C_A for each element can be determined

$$C_A = \frac{N_A}{\sum_i N_i} = \frac{I_A / [S_A \lambda(E_A)]}{\sum_i I_i / [S_i \lambda(E_i)]}$$



Example: stoichiometry of SiO₂ surface



$$\frac{I_{\text{Si}2p}}{I_{\text{O}1s}} = \frac{S_{\text{Si}2p} N_{\text{Si}} \lambda(\text{Si}2p)}{S_{\text{O}1s} N_{\text{O}} \lambda(\text{O}1s)} = 0,14$$

$$\frac{S_{\text{O}1s}}{S_{\text{Si}2p}} = 5,2 \quad \frac{\lambda(\text{O}1s)}{\lambda(\text{Si}2p)} = 0,76$$

$$[\text{Si}]/[\text{O}] = 0,14 \times 5,2 \times 0,76 = 0,55$$

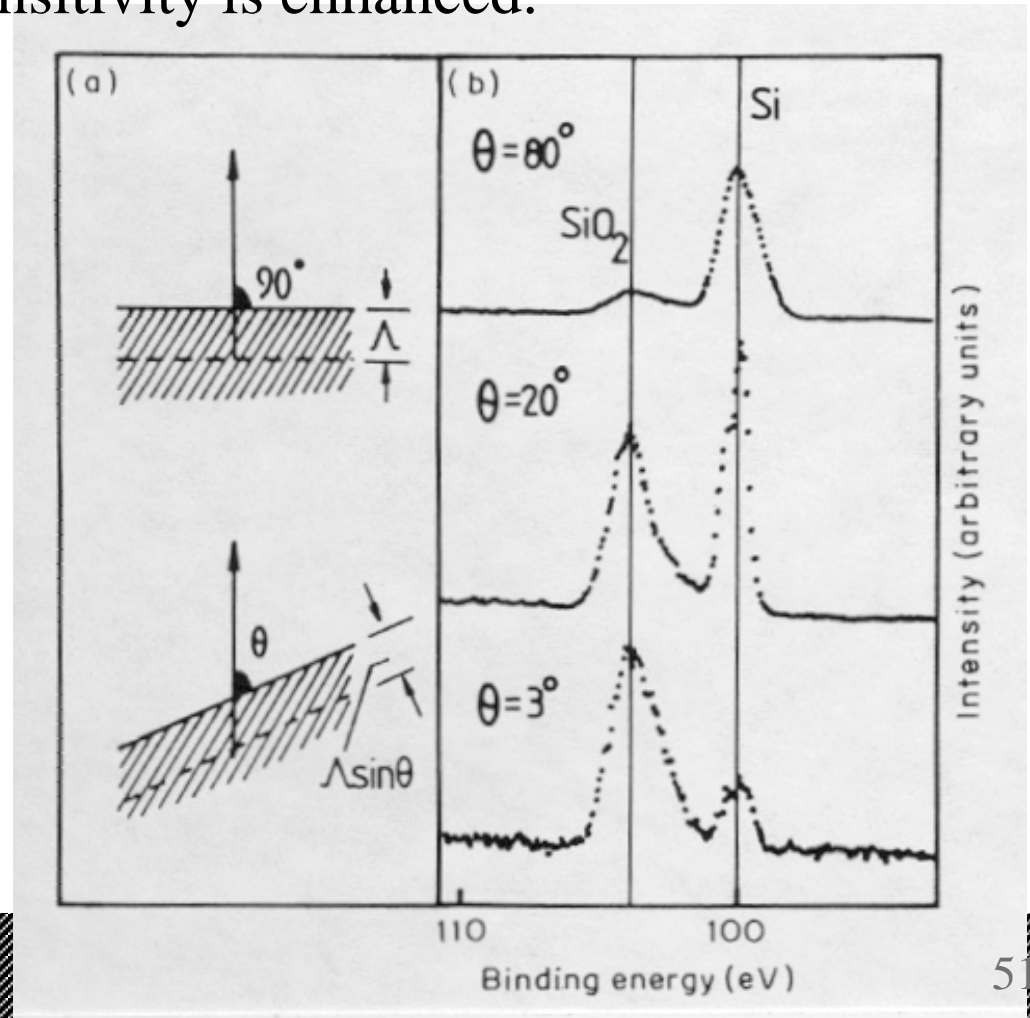


XPS : depth profiling via angular dependence

By changing experimental geometry from normal emission to grazing emission surface sensitivity is enhanced.

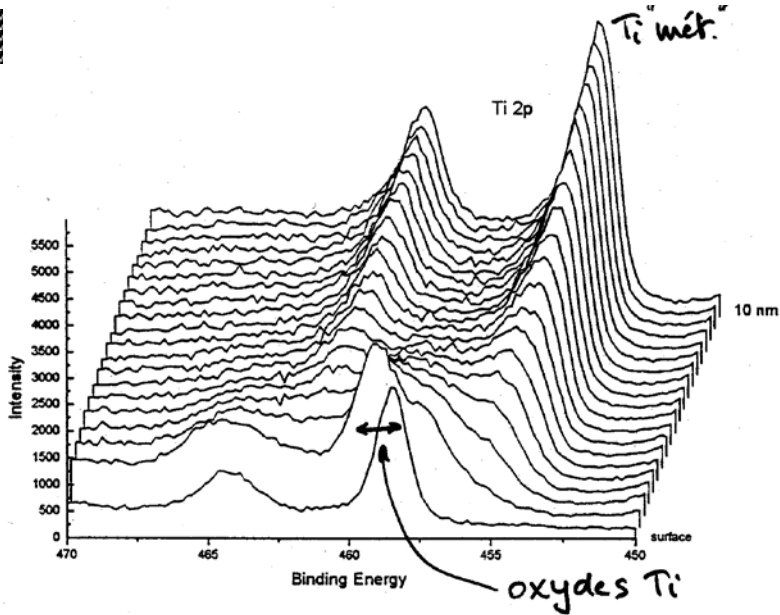
Example:

oxidised Si substrate



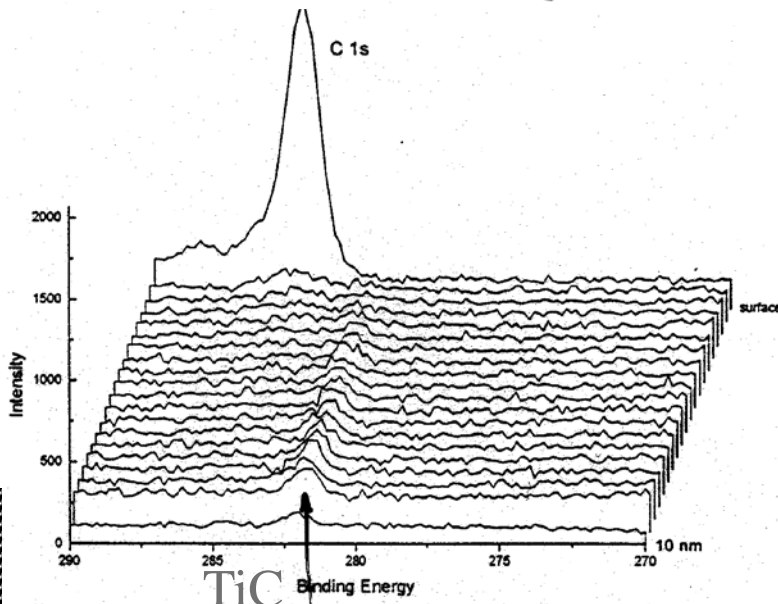


XPS : depth profiling via sputtering and measuring cycles - Destructive!



- ion sputtering - removes surface layers
- depth calibrated with reference sample

Ti 2p region

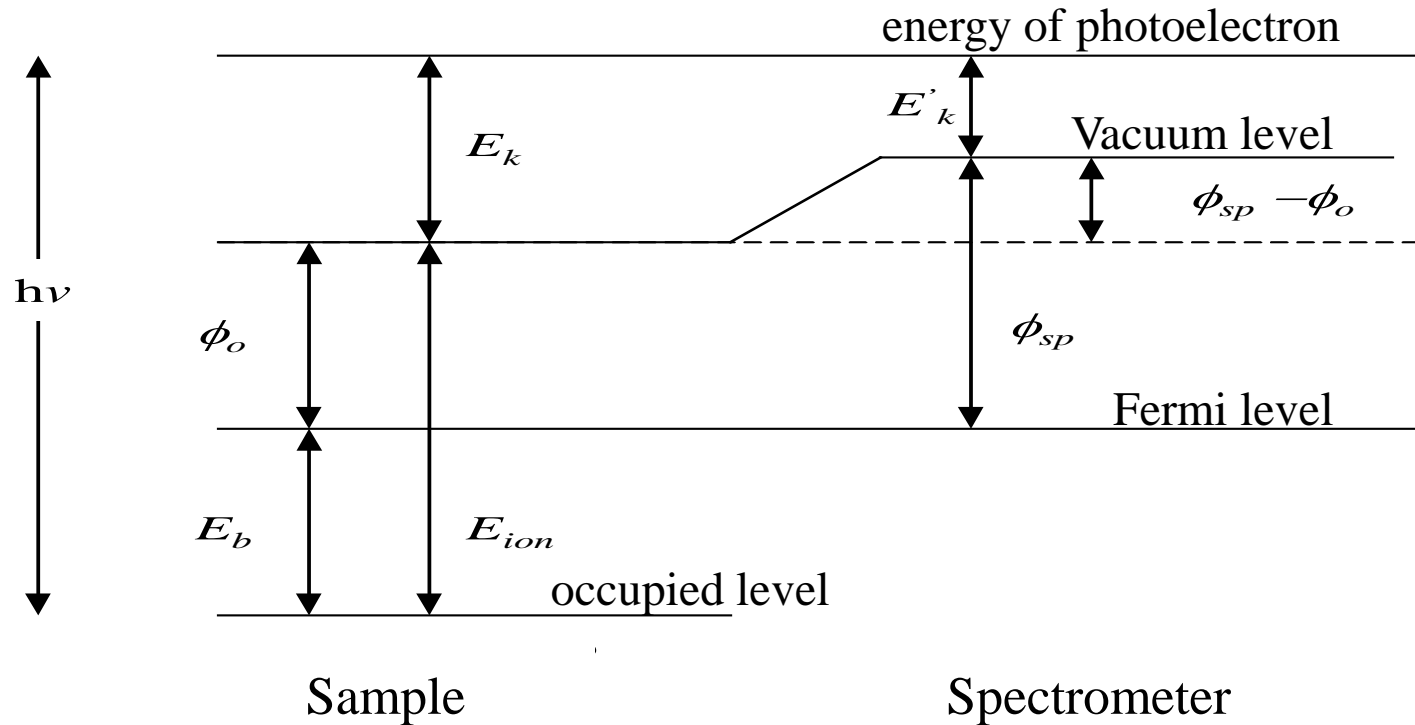


C 1s region

Example: Dental implants, surface oxidized to create porous layer which favours bone adhesion. Can determine thickness of oxide layer; can show hydrocarbon contamination at surface and carbide contamination in bulk



What is the binding energy in XPS ?



$$h\nu = E_b + E_k + \phi_o$$

$$E_b = E_f - E_i = h\nu - E'_k + \phi_{sp}$$



Energy reference

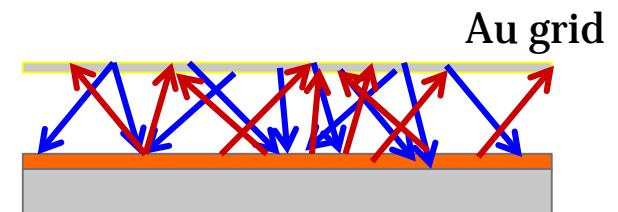
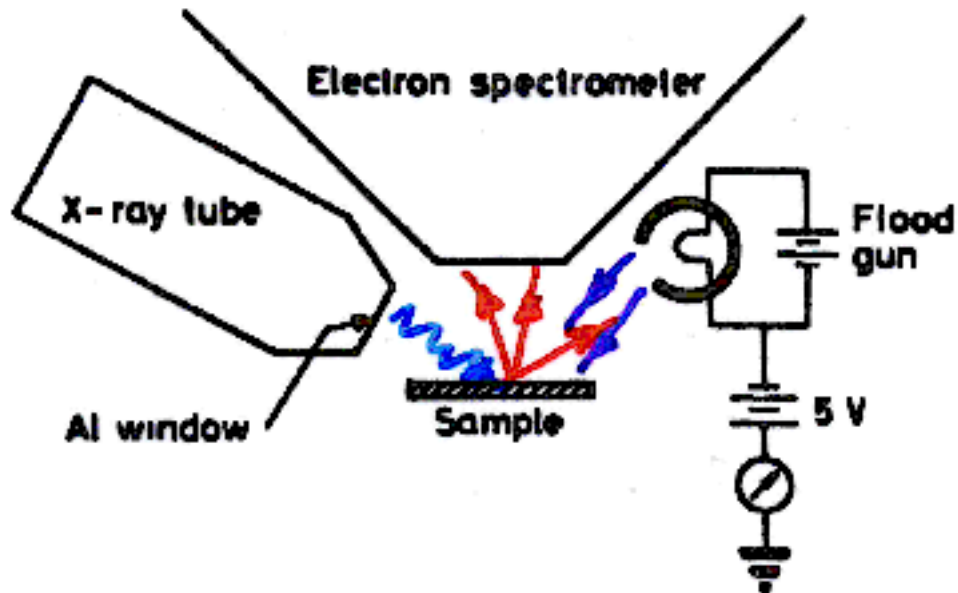
conducting samples : Fermi level

- non-conducting samples : Vacuum level

- insulating samples \Rightarrow charge effect

- $E_b = h\nu + E'_k - \phi_{sp} + \Delta E$

sample has to be flooded with low energy electrons.



Photoelectrons
generate secondary
electrons which
compensate charge



Some applications

- **quantify surface stoichiometry**
- **verify valence state of metal of nanoparticles**
- **verify core/shell structure**
- **determine growth mode of thin film**
- **verify in-situ ion transport (battery)**



High kinetic energy XPS

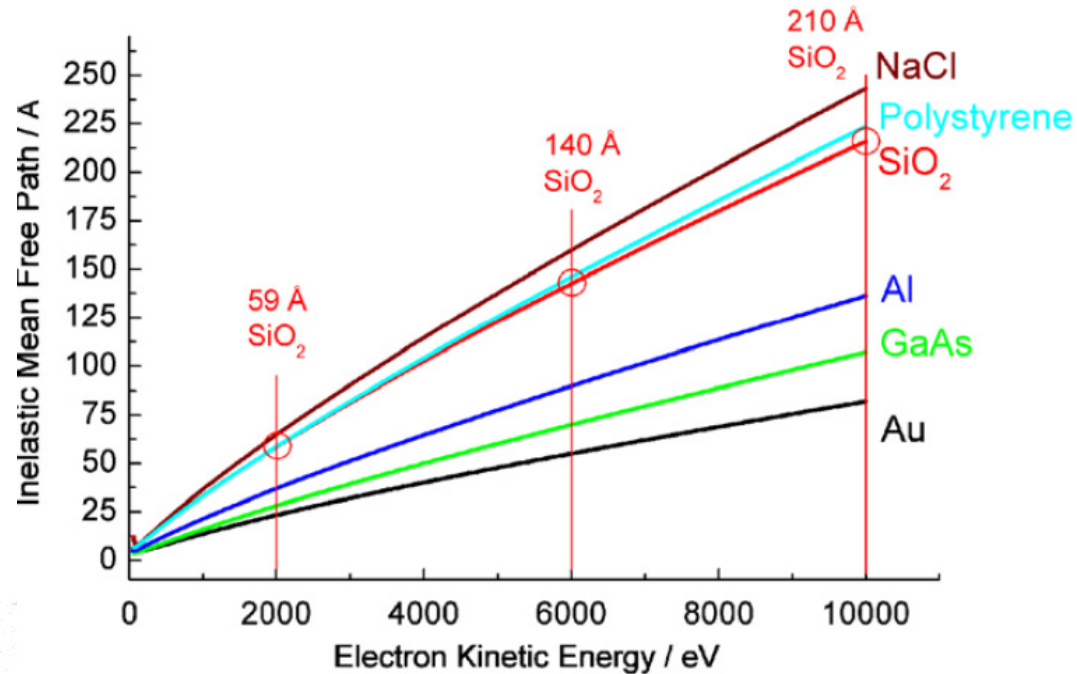
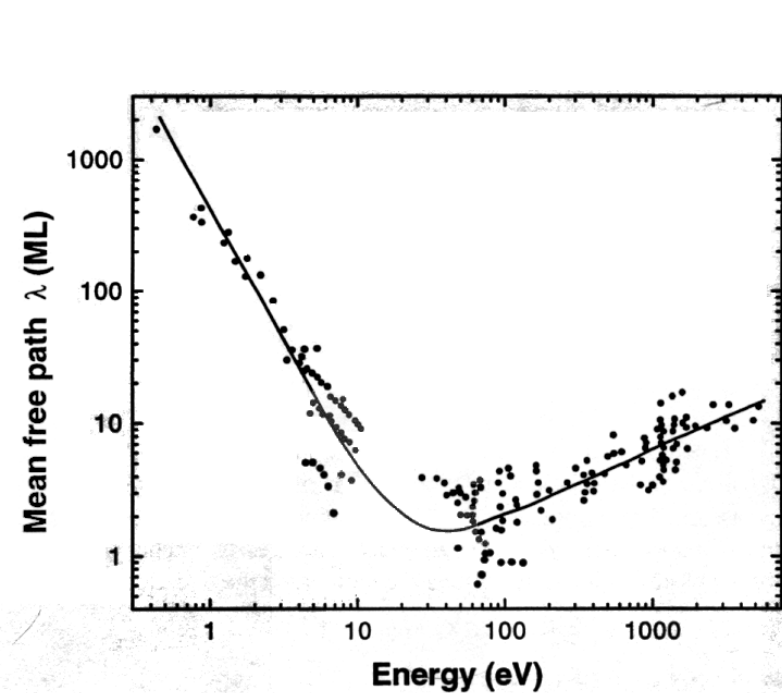
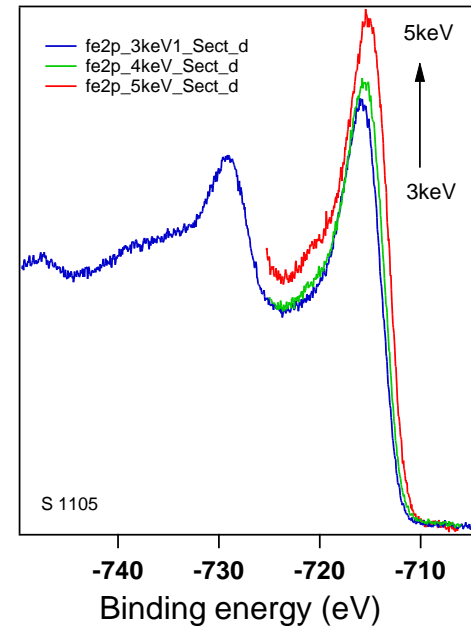
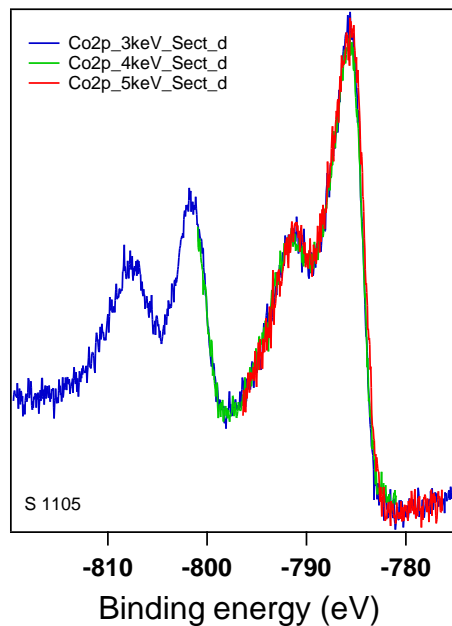
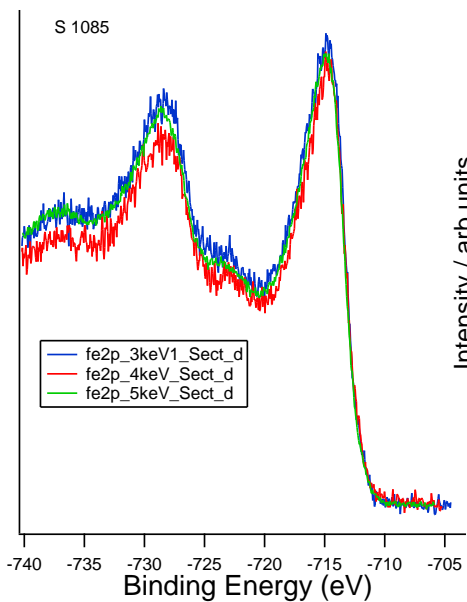
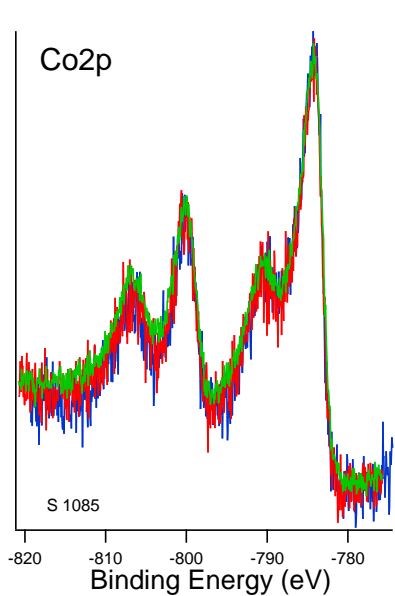
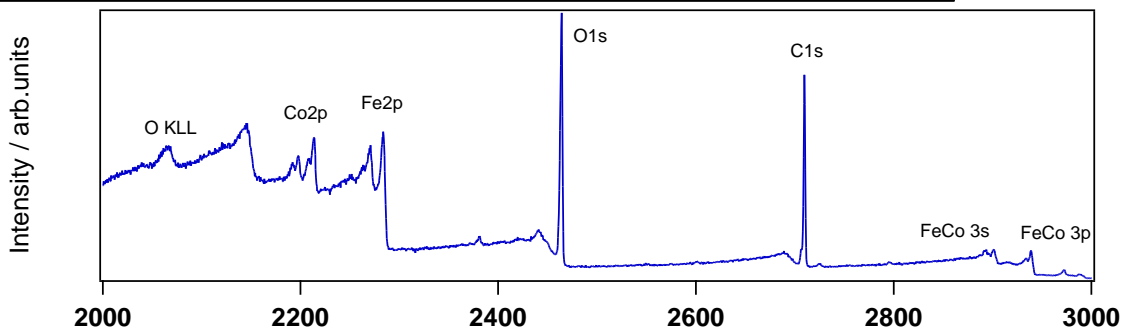


Fig. 1. Inelastic mean free path as function of electron kinetic energy for selected materials. Calculation using the NIST electron inelastic mean-free-path database, version 1.1 [3]. The HIKE facility operates between 2 and 10 keV.



FeCoO/FeO shell-core nanoparticles



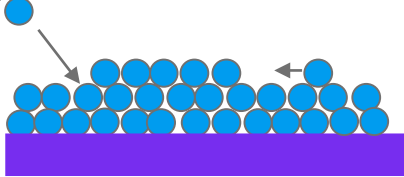
Sample A: No shell-core structure

Sample B: shows the expected shell-core structure. Co²⁺ and Fe²⁺ / ³⁺ are present in the shell and a supposed Fe³⁺ phase emerge when probing deeper

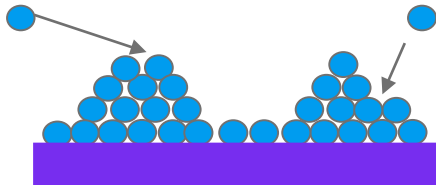


With XPS we can determine the growth mode of a film

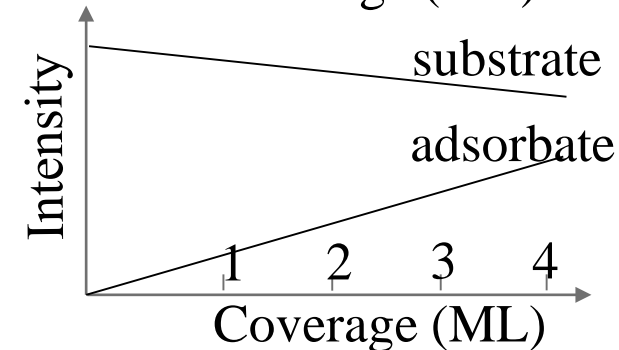
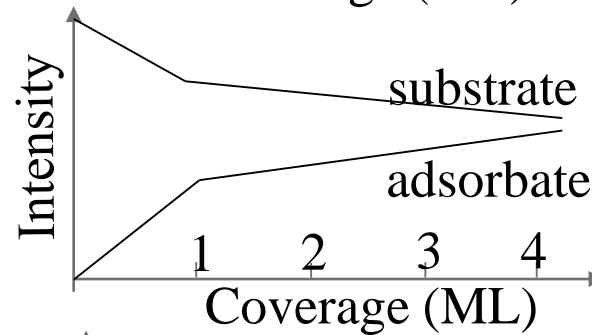
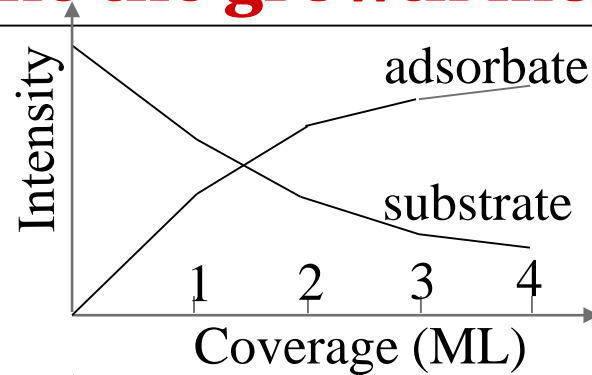
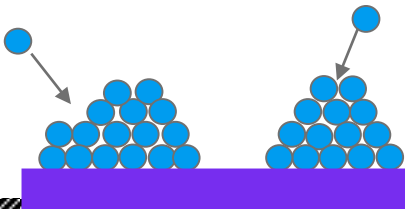
Layer by layer growth (Franck-van der Merwe)



Layer plus island growth (Stransky Krastanov)

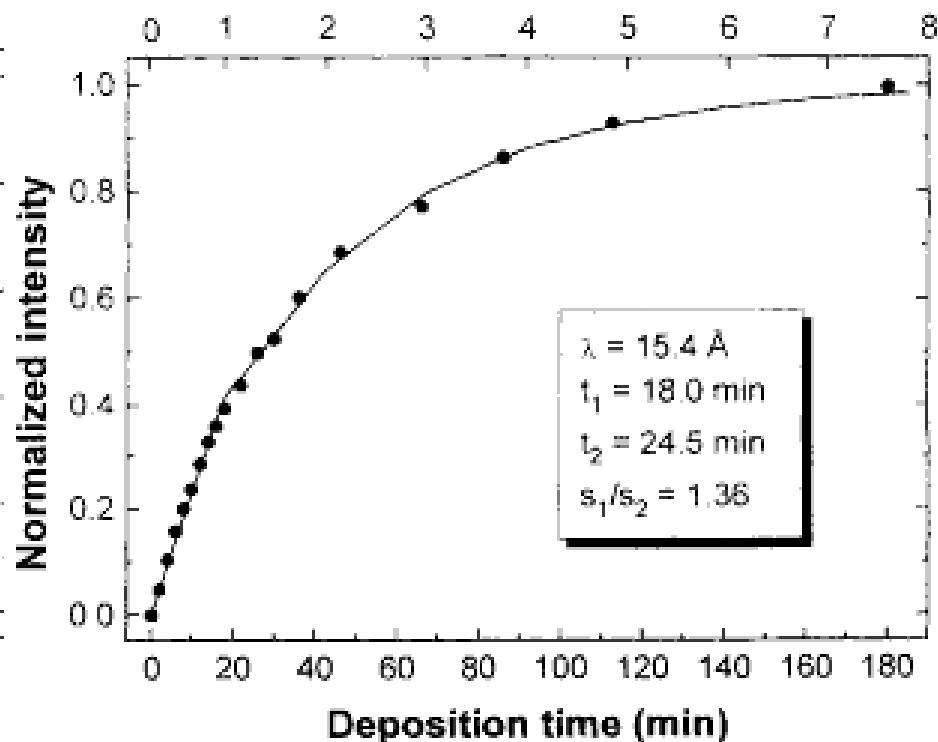
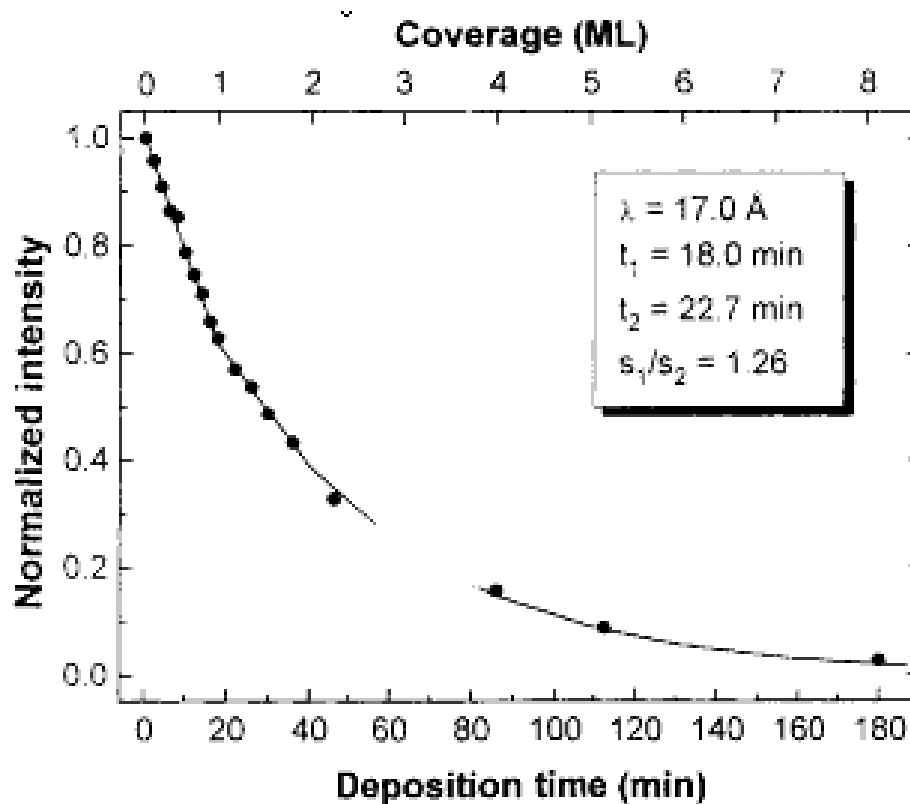


Island growth (Vollmer-Weber)





Film growth mode seen in photoemission: $C_{60}(111)/GeS(001)$





Operando XPS of solid electrolyte interphase formation and evolution in $\text{Li}_2\text{S-P}_2\text{S}_5$ solid-state electrolytes

K.N. Wood *et al.* Nature Commun. (2018) 9, 2490

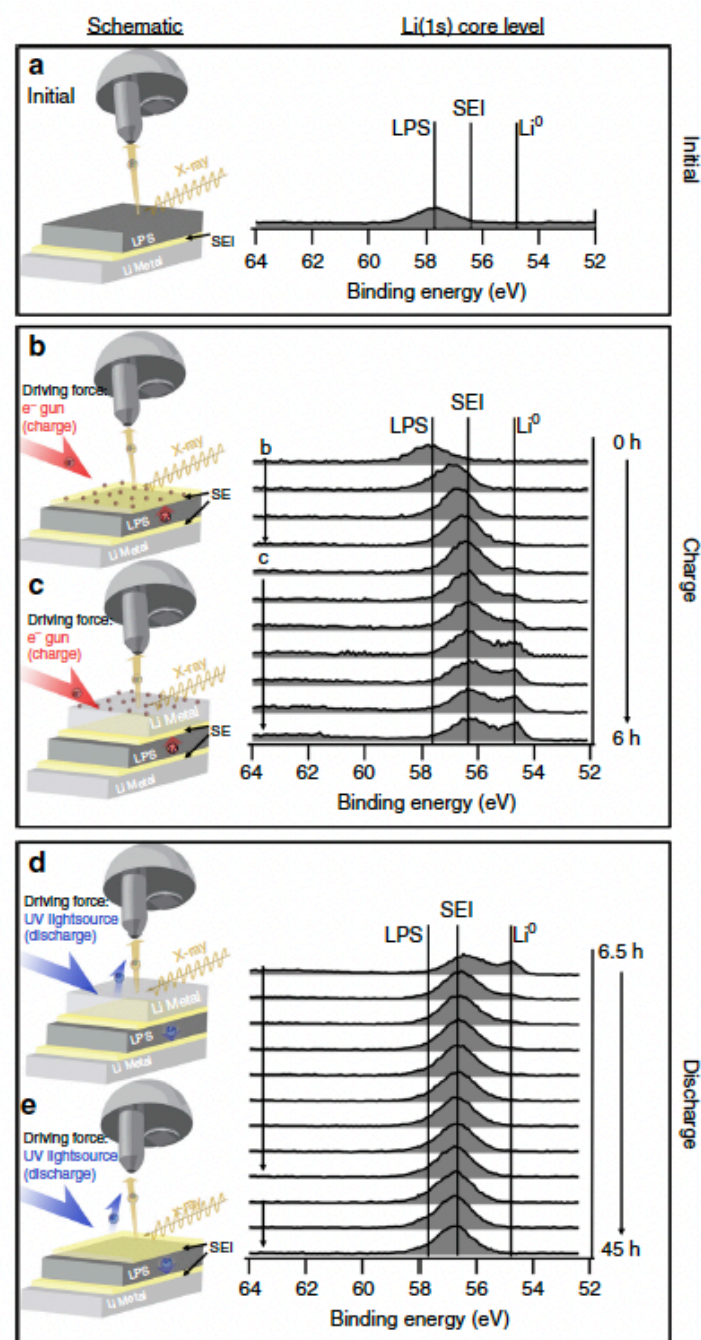


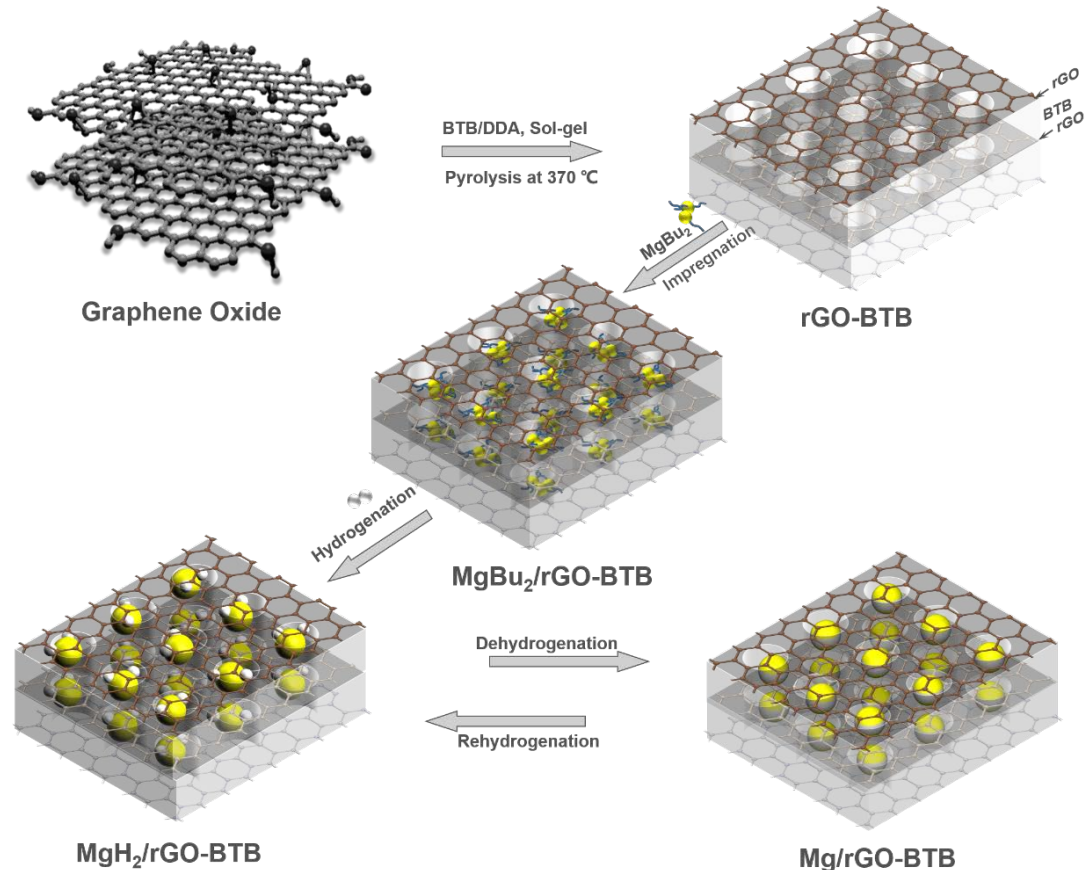
Fig. 2 opXPS schematic and Li 1s core level evolution. **a** Clean LPS surface, **b**, **c** SEI formation during charging, and **d**, **e** SEI evolution during discharging



MgH₂ nanoparticles confined in reduced graphene oxide pillared with organosilica: a novel type of hydrogen storage material

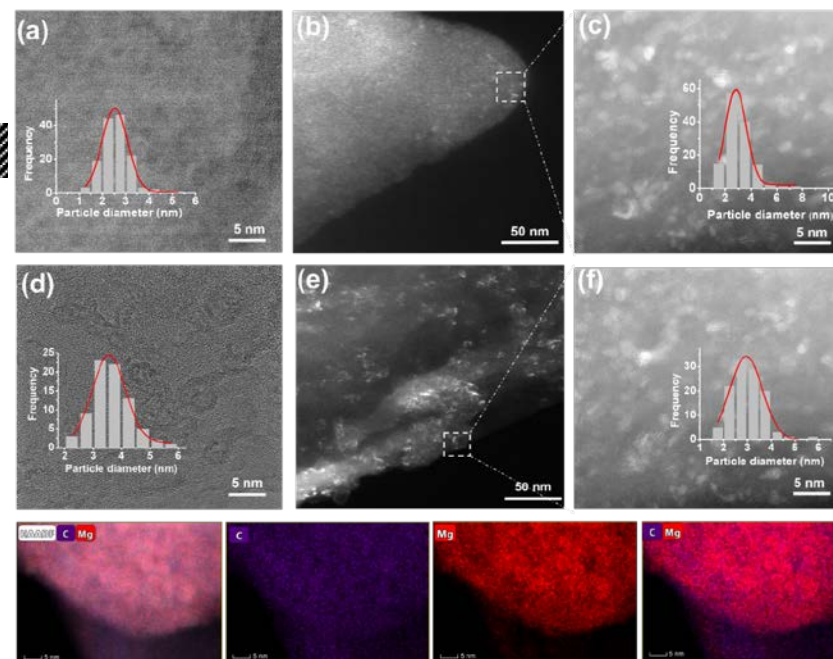
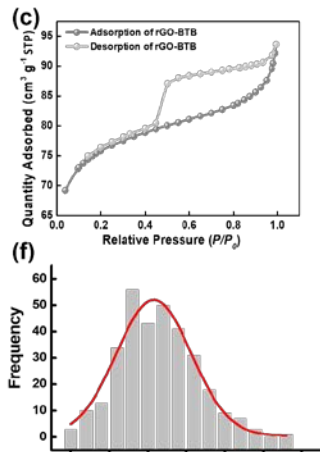
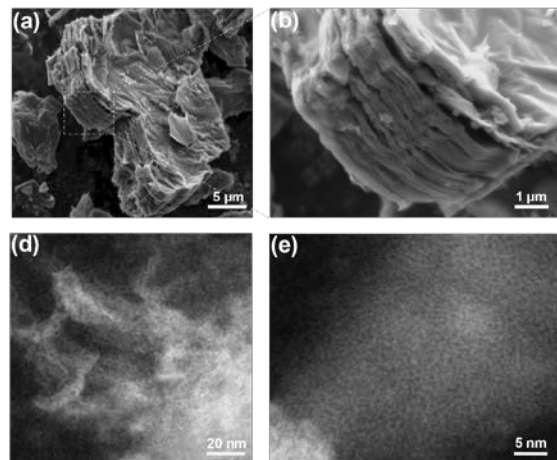
1, 4-bis(triethoxysilyl)benzene (BTB) –
pillars to distance GO planes

di-n-butylmagnesium (MgBu₂)

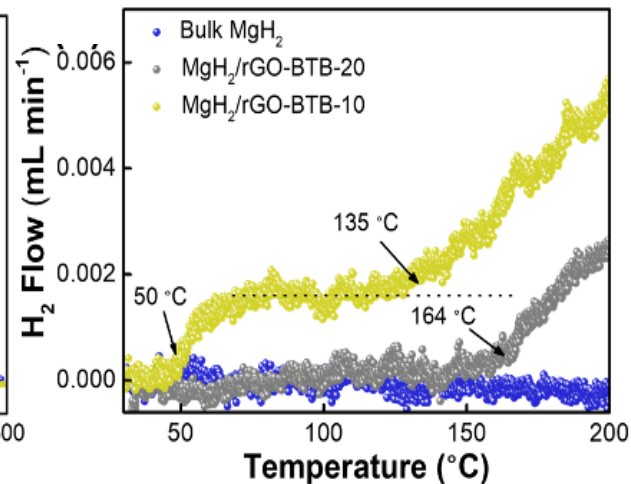
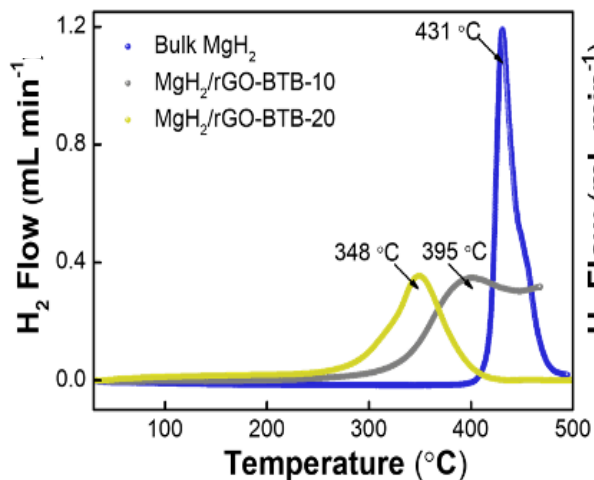


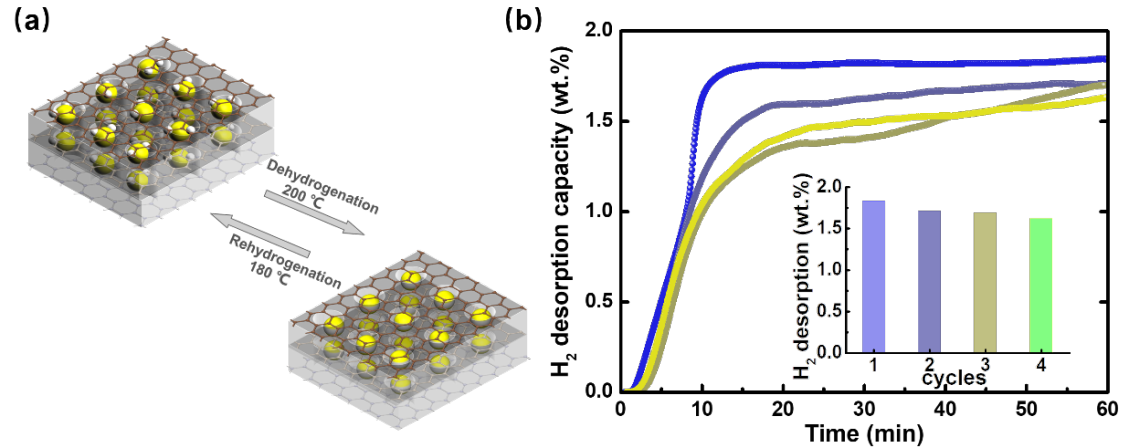


rGO-BTB-10

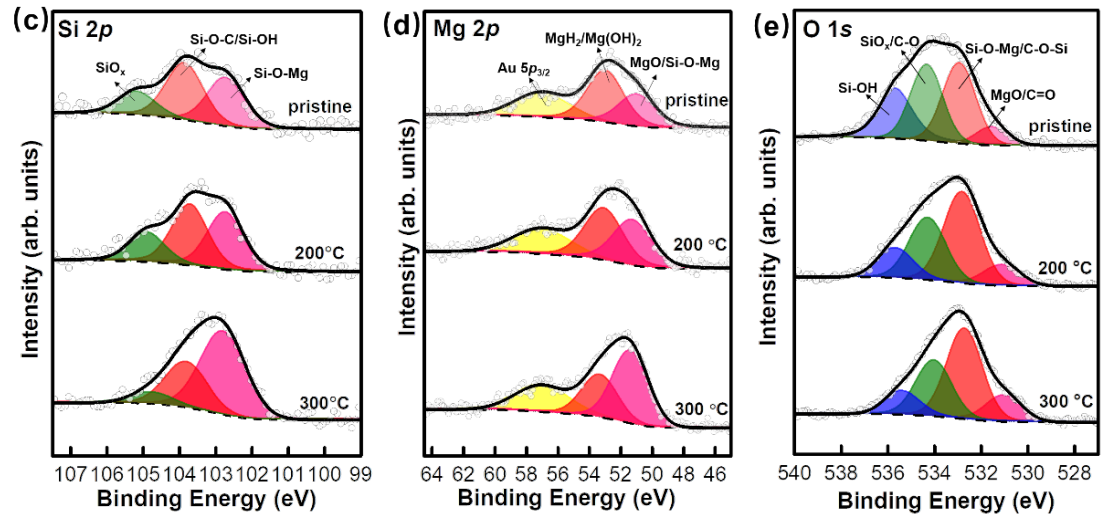


Hydrogen
desorption starts
at 50 °C ! efficient
reversible 1.62 wt.%
hydrogen storage can
be realized at 200 °C





**Above 200 °C
Mg reacts with
silanol groups**





XPS summary

- all kinds of samples: solid, liquid, gases in vacuum (10^{-8} torr)
- both insulating and conducting samples
- can detect most elements at the 0.1% level (not H, He)
- non-destructive (X-ray beam damage possible)
- surface sensitive
- quantitative analysis possible
- determination of oxidation state possible
- depth profiling : through angular dependence (non-destructive) or sputtering-measuring cycles (destructive)



university of
 groningen

Thank you for your
 attention!

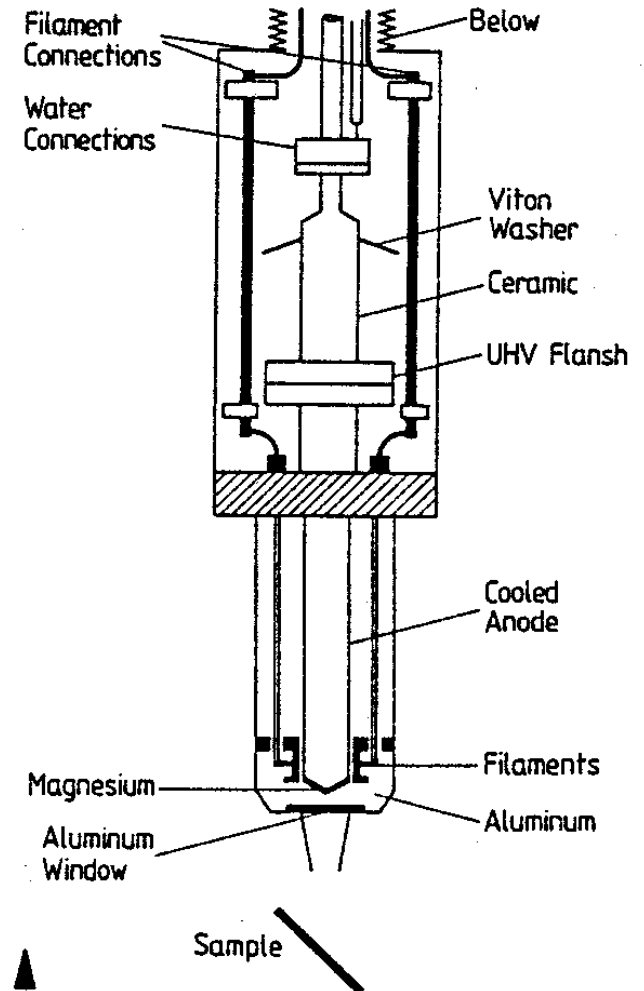


Experimental details

X-ray Sources :

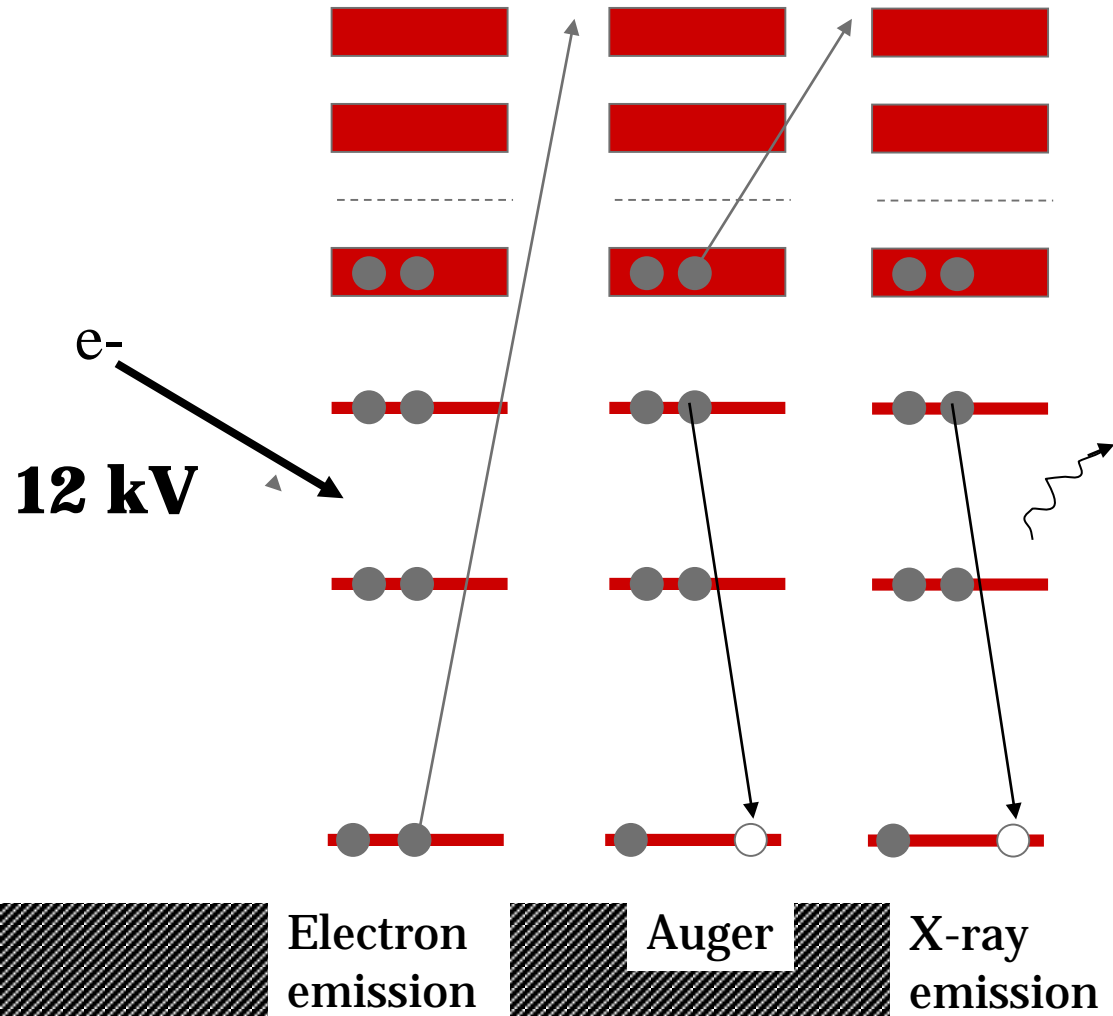
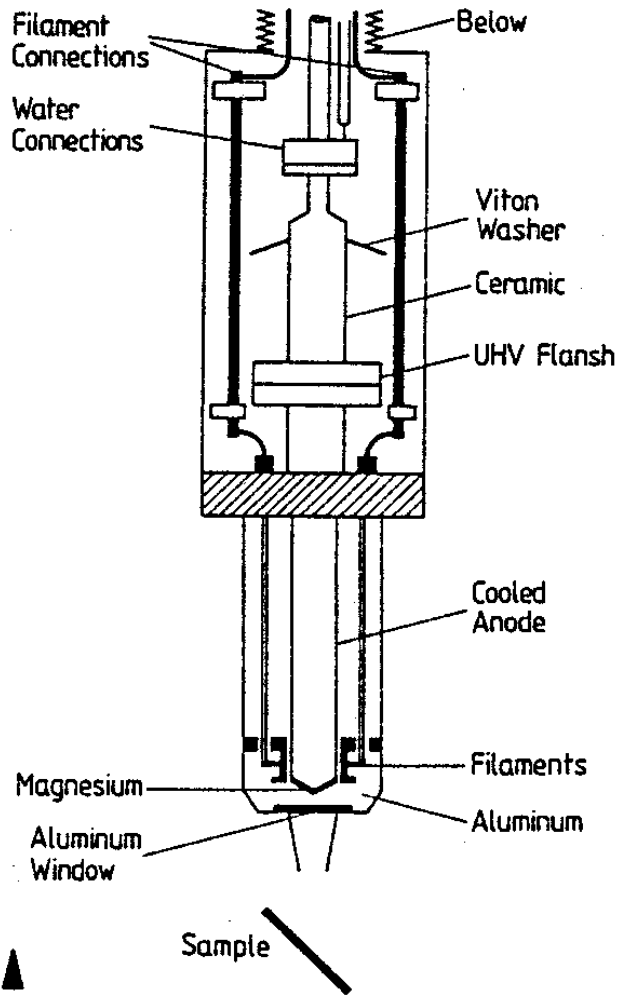
- Monochromatic laboratory sources
Mg K_{α} 1253,6 eV or Al K_{α} 1486,6 eV
- synchrotron radiation \Rightarrow photon energy
can be chosen between 200 and 1500 eV

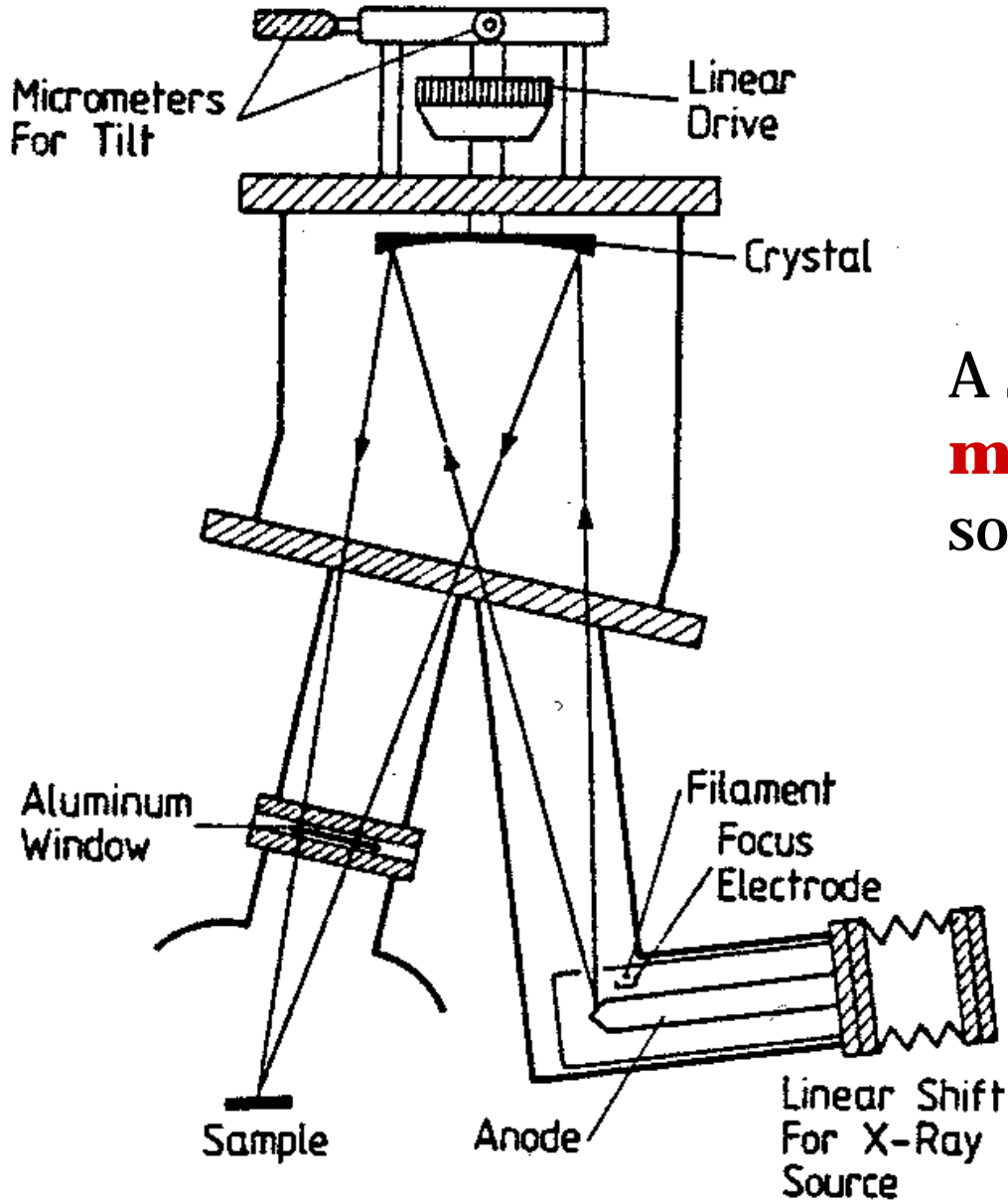
A standard **X-Ray source** –
has to be coupled to
monochromator



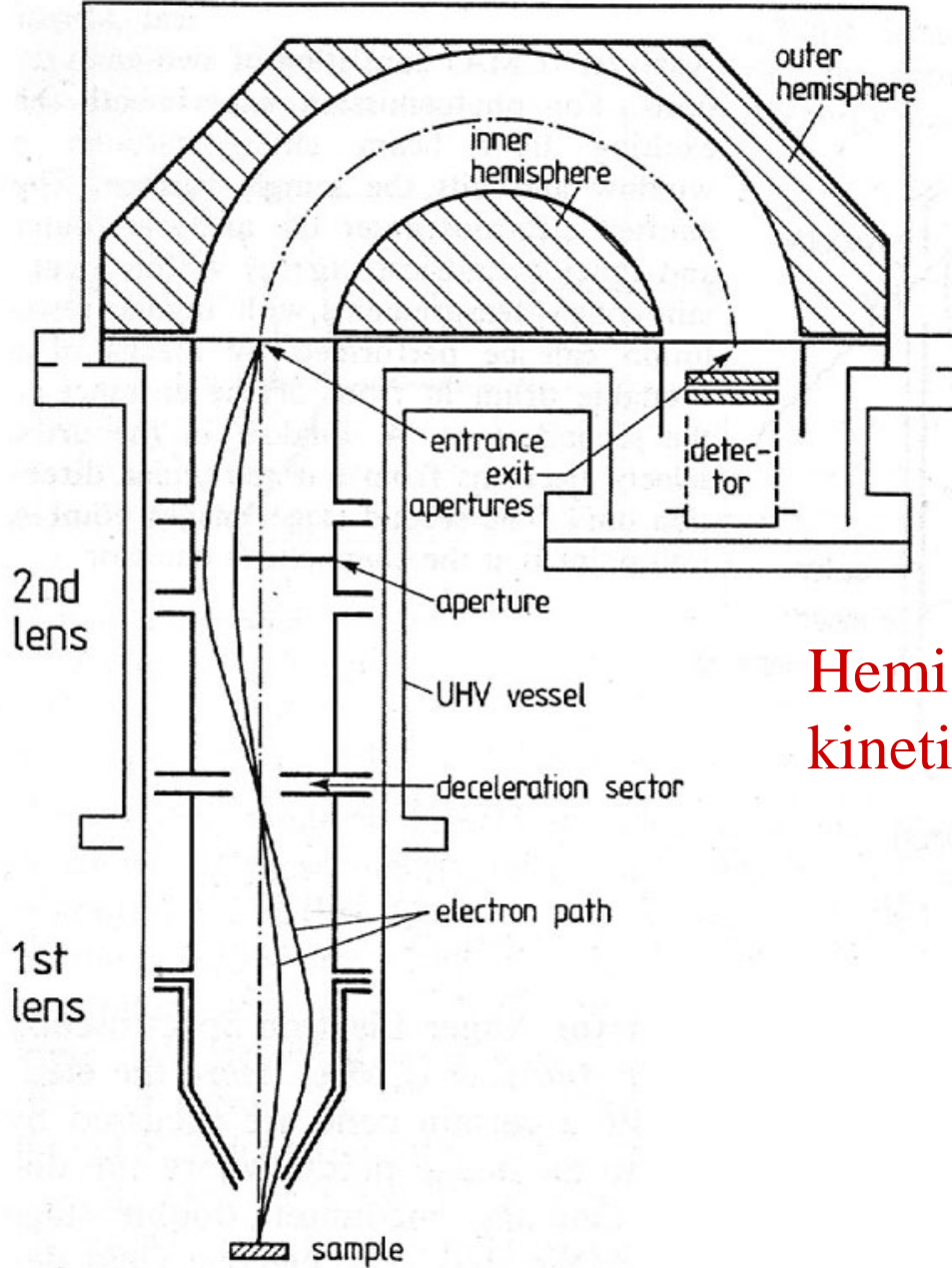


X-Ray source





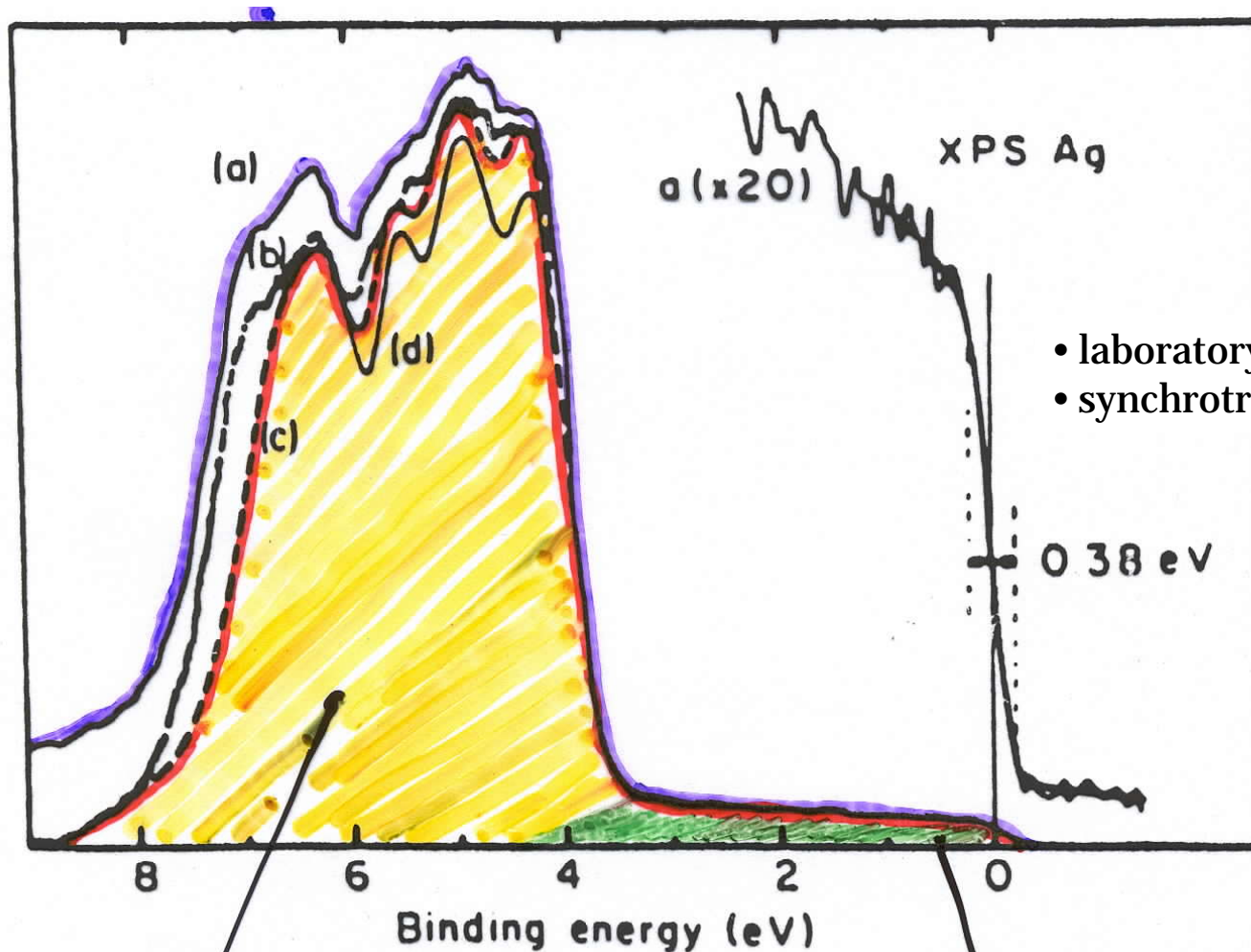
A standard
monochromator for X-Ray
source



Hemispherical analyzer of electron kinetic energy



The **instrumental resolution** can be determined by measuring the Fermi edge of a metallic sample



- laboratory sources 0,3 - 1,2 eV
- synchrotron radiation < 0,1 eV

0.38 eV

Binding energy (eV)

Ag 4d-related manifold

Ag 5(sp)-related manifold